

20.4-01 SURSTRUCTURES ET INCOMMENSURABILITE DANS LA THIOUREE. By A. Moudden, F. Denoyer, M. Lambert, Université de Paris Sud, Laboratoire de Physique des Solides Bâtiment 510, 91405 Orsay, France.

Nous discuterons le diagramme de phase P.T de la thiourée $SC(ND_2)_2$, nous montrerons que la phase modulée se subdivise en plusieurs surstructures commensurables simples 3, 7, 9 ...

Nous présenterons qualitativement le mécanisme des accrochages de la période de surstructure aux multiples impairs de la période du réseau moyen. A partir des règles de sélection des réflexions satellites nous discuterons la valeur du déphasage entre sous-réseaux et déduire le super groupe d'espace possible lorsque la période de modulation est incommensurable et voir sa compatibilité avec le groupe d'espace tridimensionnel habituel lorsque la période est commensurable. Nous analyserons, ensuite, les variations de l'amplitude de l'onde de modulation et de ses harmoniques ; nous déduirons des facteurs Debye-Waller dits anormaux la contribution des défauts locaux de la période que nous comparerons à la théorie des fluctuations de phase des structures modulées.

20.4-02 $Er_{1.2}Mo_6S_8$ --STRUCTURE WITH THREE INCOMMENSURATE SUPERSTRUCTURES AND DISORDER. H. Brigitte Krause, Physics Dept., Northern Ill. U., DeKalb, IL, 60115, USA, and John M. Cowley, Physics Dept., Arizona State U., Tempe, AZ, 85281, USA.

A series of $(Sn Er_{1-x})_{1.2}Mo_6S_8$ compounds was examined with selected area electron diffraction and high resolution electron microscopy. In addition to the expected rhombohedral (approximately cubic) Chevrel phase, several different superlattices and sublattices were observed. Furthermore, for $Er_{1.2}Mo_6S_8$ a new monoclinic, or nearly monoclinic, structure type was discovered. The b-direction of the monoclinic unit cell coincided with the b-direction of the pseudocubic Chevrel phase, and the lattice constant $b_C = b_M = 6.37\text{\AA}$ is identical for both phases. The monoclinic a-axis is $a_M = 12.10\text{\AA}$, the c-axis is $c_M = 8.17\text{\AA}$, and the angle $\beta_M = 96.5^\circ$. The monoclinic axes are rotated with respect to the pseudocubic axes such that $[101]_M$ is in the direction of $[001]_C$ and $[10\bar{1}]_M$ approximately in the direction of $[100]_C$.

In spite of the similarities between the cubic and monoclinic geometry, there are distinct differences: the volume ratio of $\frac{V_M}{V_C} = 2.4$ does not allow a simple interpretation in terms of an Mo_6S_8 cube arrangement as in the Chevrel phase. In addition to regular monoclinic reflections, incommensurate reflections and diffuse scattering were observed. The diffuse scattering occurred in reciprocal lattice sheets perpendicular to the b-direction. One of the incommensurate doublets occurred at $[110]_M$

with a splitting parallel to the b-axis and corresponding to about 100\AA in real space. Another doublet at $[010]$ had splitting in the cubic $[101]$ direction. A third series of incommensurate reflections occurred in the cubic $[110]$ direction with splittings both in the direction of $[110]_C$ and $[010]_C$. The incommensurate reflections and lattice images indicate a modulation of the structure interpretable as antiphase domains. Structure models will be proposed.

20.4-03 THE MODULATED STRUCTURE OF CUBIC PHASE Cu_9BiS_6 . By K. Tomeoka and M. Ohmura*, Dept. of

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Quenched single crystals synthesized at about 550°C for compositions near Cu_9BiS_6 were found to reveal non-integral type satellite reflections in the X-ray diffraction patterns. The crystals reveal fundamental reflections corresponding to cubic α -chalcocite, which has $a = 5.54\text{--}5.57\text{\AA}$ cell edge and a face-centered cubic cell. Around the fundamental reflections, many non-integral type extra reflections occur in clusters. These are described as follows: (I) The period of the satellite reflections corresponds to a non-integral multiple of the subcell and changes in the range of 6.0 to $6.75a$. (II) The satellites appear three-dimensionally and make a body-centered cubic reciprocal lattice in the same manner as the main reflections. (III) Generally, intensities of the satellites are very strong, and some of them are much stronger than the main ones. (IV) Compared with the mains, the satellites are slightly diffuse. The degree of diffuseness differs in each sample. (V) Asymmetry of intensities between pairs of satellites is very remarkable. Generally, intensities of lower angle satellites are stronger than those of higher angle ones. This cubic phase has a solid solution range of some extent. The positions and the diffuseness of the satellites are correlated with the change of composition of the crystal.

The superstructure was investigated by using X-ray diffraction intensity data, measured on a crystal having a 6.5 multiple of the subcell, because the period can be made integral (13 multiple) simply by doubling 6.5. Since the satellites form a body-centered cubic lattice

and satisfy $m\bar{3}m$ point symmetry, the superstructure was assumed to have the highest space group $Fm\bar{3}m$ for convenience of crystallographic calculations. The actual structure was examined as a superstructure having a face-centered cubic lattice with a lattice constant of $a = 13a = 72.32\text{\AA}$.

Through analysis of the partial Patterson function, the superstructure was found to be basically defined by clusters of Bi and of Cu. These two kinds of metal clusters occur alternately and are arranged in a face-centered cubic lattice mode. Such local clustering of the atoms is interpreted to be a combination of two different kinds of modulation factors: the periodic substitution between Bi and Cu and the periodic positional displacement of the atoms. Structure factor calculations, based on the clustering structure, gave substantial agreement with observed intensities.

The problems of the clustering structure can be attributed to the process of cooling from high temperature. This cubic phase and $Cu_3Bi_2S_9$, which also has a non-integral type superstructure (Ohmura, Tomeoka & Sadanaga, 1979, AIP Conf. Proc., 53, 355), are stable only at temperature higher than 400°C and metastable at room temperature. In their stable high temperature phases, the crystals exist with random replacement between Bi and Cu and have structures approximating the substructures as defined in our studies (Ohmura, 1973, Neues Jahrb. Miner. Monatsh., 227; Tomeoka, Ohmura & Sadanaga, 1980, Mineralogical Journ., 10, 57.) During quenching, however, the metal atoms segregate into two kinds of clusters. The long-range repeat distances are directly influenced by these metal clusters and seem to be independent of the period of the subcell.

20.4-04 THE ENERGETICS OF FORMATION AND ORDERING OF CRYSTALLOGRAPHIC SHEAR PLANES IN ReO_3 STRUCTURED OXIDES. By A.N. Cormack, C.R.A. Catlow, University College London, and P.W. Tasker, AERE, Harwell.

We present the results of a detailed atomistic computer simulation study of extended defect formation in ReO_3 (DO_3) structured nonstoichiometric oxides. The phenomenon of crystallographic shear (CS) planes, as extended defects, in such compounds as WO_3-x and MoO_3-x has been extensively studied by both X-ray diffraction and high resolution electron microscopy. This combination of Reciprocal Space and Real Space crystallography has shed considerable light on the geometrical structure of these defects which may be considered as perturbations of a periodic, modulated structure (Proc. Int. Conf. on 'Modulated Structures', Hawaii 1979. American Institute of Physics conf. proc. No. 53, Plenum Press, New York.). At the same time, however, several fundamental problems have been raised concerning the preferred orientation for the formation of isolated CS planes, and their long range ordering into regular, or nearly regular, arrays at greater concentrations in the crystal. Our model correctly establishes that (102) CS planes are the most energetically favoured isolated CS planes in ReO_3 type structures and we also show that lattice relaxations play an essential part in stabilising this orientation over others such as (001) and (103). However, the energies of these defects are sufficiently close, so that the difference between them in the isolated case becomes outweighed by terms arising from their interaction at larger concentrations with increasing

departure from stoichiometry. We also establish that at very low deviations from stoichiometry, dissociation of CS planes into anion vacancies may be possible due to the closeness in energy of the point defect mode of reduction with the extended defect mode. From our simulations we have thus been able to rationalise the apparently complex relationship between structure and composition in these nonstoichiometric oxides.

20.4-05 IMPLICATIONS OF PSEUDO SYSTEMATIC ABSENCES IN APPARENT TETRAGONAL $4/mmm$ DIFFRACTION SYMMETRY. By A.D. Rae, School of Chemistry, University of New South Wales, Kensington, 2033, N.S.W. Australia.

The absences $F(hh\ell) = 0$, $h + \ell = 2n + 1$ and $F(hh\ell) = 0$, $h = 2n + 1$ do not correspond to symmetry elements of any primitive space group of $4/mmm$ Laue symmetry. When either systematic absence occurs in data of apparent $4/mmm$ diffraction symmetry, it may imply that the true crystal structure is not $4/mmm$ Laue symmetry but is of either mmm or $4/m$ Laue symmetry. Diffraction data with $\{hk\ell\}$ strong only for $h + k = 2n$ may be described as arising from a disordered structure in a subcell of axial lengths $a/\sqrt{2}$, $a/\sqrt{2}$, c which may indeed be of $4/mmm$ Laue symmetry. Systematic absences of $F(0k'\ell') = 0$ for $k' + \ell' = 2n + 1$ or $F(0k'\ell')$ for $k' = 2n + 1$ correspond to an allowed symmetry element for this disordered subcell structure. Twinning must be a condition for $\{hk\ell\}$ data with $h + k = 2n + 1$ to also have $4/mmm$ diffraction symmetry, should a satisfactory disordered structure consistent with the systematic absences of the subcell be obtained. Two structures based upon spacegroup $P4_2/mmm$ have been refined, one of true spacegroup $P4_2/n$, the other of true spacegroup $Pbam$. Both showed $|F(hk\ell)| = |F(kh\ell)|$ for all data. Constrained refinement is essential when twinning occurs. Fourier analysis methods identify which combinations of parameters are poorly defined.