tions to the $I$, by comparing the XRDS intensity at equivalent points in reciprocal space. On the contrary, scattering contributions due to static displacements of the atoms were neglected, since they were evaluated as being on the order of a few 0.1% of the measured intensity. As a result, the XRDS data were well fitted by the monoclinic Von Tscharner scattering function $I(x) = (e^{-x} - 4\ell) / (1-x)$ with $a$ and $b$, as scattering centers of Ga and Al atoms, respectively. This result shows that Ga and Al atoms, in 1% GaAs and InGaAs, are distributed in the cation FCC sub-lattice in a completely random way. Moreover, the possibility of using XRDS experiments to investigate the atomic arrangement within the substituted sub-lattice in III-V solid solutions has been demonstrated. Recent investigations of phase separation and clustering criteria (see, e.g., Stringfellow G.B., J. Electronic Materials 11, 903 (1982)) indicate that XRDS investigations on pseudobinary alloy semiconductors, such as InGaAs and InGaAs, can of great interest.

11. S-8 DEFECT STRUCTURE OF DEUTERIUM IN NIOBiUM - A NEW NEUTRON SCATTERING STUDY. By H. Dorsch, J. Peisi, Sektion Physik, Universität München, Germany; E. Burkel, Cornell University, Ithaca, N.Y. 14853, USA; B. Dorner ILL, Grenoble, France.

Detailed information on the location of deuterium, the long ranged displacement field and the local atomic distortions of the niobium lattice is deduced from measurements of the coherent elastic diffuse neutron scattering intensity distribution. Huang diffuse scattering (measured with D10 of the ILL) confirms the cubic symmetry of the long range displacement field and the elastic force dipole tensor $P_{ij} = 4\delta_{ij} (3.52 \pm 0.05)$ eV.

The asymmetry of the scattering distribution leads to a location on tetrahedral sites. The scattering at large Q far away from Bragg reflections, the so-called "Zwischenreflex" - scattering (measured with IN2 of the ILL), cannot be explained by the model deduced from Huang diffuse scattering results. Model calculations which can explain both lead to a local defect structure, where lattice relaxation effects due to the rapid motion of deuterium in the lattice have to be taken into account. This also supplies an explanation for the cubic symmetry of the long ranged displacement field of a defect on tetrahedral sites.

11. S-1 X-RAY POLARIZATION BY BRAGG DIFFRACTION FROM BENT AND FLAT CRYSTALS. By J. Zahr, Chemistry Dept., Northern Arizona University, Flagstaff, Az 86011, USA.

Recently much attention has been given to the utilization of polarized X-rays in X-ray fluorescence spectrometers (R. Ayon, et al., Adv. in X-Ray Anal. 15, 11-24 (1982)). The introduction of the extra scattering event to polarize the beam decreases the intensity. 90° scattering from the interior of a perfect crystal reduces the intensity by utilizing a manifold of beams. This paper concerns itself with the development of a phenomenological theory, based on the mosaic model, to serve as a guide to developing better X-ray optics for Bragg scattered polarized X-ray spectrometers. The improved efficiency of the bent crystal (Johann) over the flat crystal has two sources. One is purely geometric and depends on the collimator(s) length $l$, and radius $r$, the radius of the Rowland circle $d$, and the displacement of the isotropic source from the Rowland circle $e$. The other source of improvement arises from the mosaic block size $s$, and from the mosaic distribution function $g$ assumed to be Gaussian. The mosaic deviation $m$ is given by:

$$m = \frac{1}{\sqrt{\pi}} e^{-s^2}$$

In the approximation of negligible true absorption and $s^2$ and $s^2$ extinction the theoretical results depend only upon geometry.

For Cu Kα diffraction from Cu113 and $l=1\text{cm}$, $e=0.8\text{cm}$, $R=1.5\text{cm}$ and $\theta=0.15\text{cm}$ the reported efficiency ratio is $bent/flat=3$ (F. Houbautchev, personal communication). Using these parameters and $l$ and $e$ equal to 0.001cm and 0.001 rad (flat) and 0.0095cm and 0.002 rad (bent) I calculate the ratio to be 3.3. Taking the t's 2 times larger and the m's 2 times larger gives 4.2. Taking $l=\infty$ and the first set of crystal parameters gives 3.9.

I hope these results will be an aid in the construction of better spectrometers and encourage the determination of the mosaic parameters.


It has been shown that an X-ray interferometer consisting of two crystals cut from different silicon materials can be successfully operated. Experiments of this kind are of particular interest if lattice spacings of different crystals are to be measured on the meter scale, or if more space in the interfering beam paths is needed. Each of the two crystals shown in the figure belongs to a complete Laue-case interferometer tested separately in order to measure the homogeneity of the crystal lattice. The first crystal, bearing both the beam splitter S and the mirror M, was part of the scanning X-ray interferometer used for the absolute determination of the (220) lattice plane spacing (Becker, Dorenwendt, Ebeling, Lauer, Lucas, Probst, Rademacher, Reim, Seyfried, Siegert, Phys. Rev. Lett. (1981) 46, 1540). The second crystal with the analyzer A belongs to an interferometer cut in a similar way for the same purpose. The relative difference in the Bragg-plane spacings of the two silicon materials was about (3 ± 1) $\times 10^{-4}$ measured by crystal-to-crystal comparison experiments (Becker, Seyfried, Siegert, Z. Phys. (1982) B 68, 17).

The geometrical deviations of the interferometer from the ideal shape caused by the manufacturing process are thoroughly investigated. In order to align the lattice planes of the two crystals parallel to one another by light optical means, the crystal surfaces are polished to form optical mirrors. Spacing marks are etched on the mirrors in order to realize equal distances between the three lamellas, S, M and A, by use of an optical length measuring device. Only a low interference contrast of $\Delta = 0.05$ was observed in the outgoing beams. The reason for this is mainly the difference in thickness of more than $100 \mu$m between the beam splitter S and the analyzer A.

11. REAL AND IDEAL CRYSTALS - C - 343