Dynamically refined PEDT of Ni2Si compared to XRD

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Electron diffraction tomography (EDT) [1] combined with the precession electron diffraction (PED) [2] are ideal techniques for structural analysis of single nanocrystals. The strong interaction between electrons and matter, compared to X-ray, allows structural analysis of nanocrystals as small as tens of nanometers. During the EDT experiment, the crystal is tilted in small steps and a diffraction pattern is collected for each position, giving three-dimensional information of the reciprocal space. However, the strong interaction between the electron beam and matter generates multiple scattering of the beam and the accurate structure refinement requires the use of dynamical diffraction theory for the calculation of model intensities. The dynamic character of the reflections can be suppressed by using PED. The technique consists of precessing the beam around the optical axis of the microscope, where the resulting intensities are integrated for each angular position of the beam. These integrated intensities are more sensitive to structure parameters and less sensitive to crystal imperfections. Despite of the damping of the dynamical effects, the refinement against precession EDT (PEDT) data using kinematical diffraction theory gives low accuracy of the structure parameters and high residual factors. The dynamical refinement for EDT data sets was implemented recently [3] and has been shown to be superior to the kinematical refinement, giving lower figures of merit and higher accuracy of the structure parameters. In this work, the dynamical refinement is used for the structure analysis of a nanowire of Ni2Si with the diameter of 16 nm. The dynamical refinement against a data set collected by PEDT is compared to kinematical refinement and to a reference structure, obtained by single crystal X-ray diffraction. The structure has orthorhombic Pnma symmetry with 3 independent atoms. It is shown that the single crystal X-ray structure and the PEDT structure match very well with an average distance of atomic positions less than 0.015 Å.


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Machine learning decomposition of scanning precession electron diffraction data

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Scanning precession electron diffraction (SPED) is being developed to map the crystalline phases in materials with nm resolution and across micron sized regions. Precession electron diffraction (PED) patterns are recorded using a double conical rocking beam system [1] as the electron beam is scanned across the sample [2]. The 4-dimensional (two real & two reciprocal) data set can be analysed in a number of ways. Most simply, ‘virtual’ dark-field images can be formed by plotting the intensity of a diffraeted beam (sub-set of pixels) as a function of beam position. Each PED pattern can also be matched to a library of simulated patterns, determining the orientation and producing orientation images, typically as Euler angle maps [2]. This analysis is hindered by ‘artefacts’ such as specimen bending and thickness variations, the effect of which can be reduced by utilising larger precession angles. The remaining challenge that we address is the appropriate treatment of diffraction patterns that comprise signals from overlapping crystals.

We apply machine learning approaches based on non-negative matrix factorisation (NMF) and principal component analysis (PCA) combined with independent component analysis (ICA) in order to learn the underlying ‘component patterns’, that make up the data, together with their associated ‘loadings’ at each real space pixel. An example of such decomposition is shown in the figure below. Loading maps indicate the region where the component pattern is significant and resembles a simplified dark-field image. Loading maps corresponding to different component patterns may have overlapping regions of intensity where mixed diffraction patterns have been separated into individual components. This decomposition is the key to performing 3D volume reconstructions with crystallographic information at each voxel [3]. Here we report on the development of this approach by evaluating the decomposition of PED data systematically using a number of algorithms. In particular, results are presented demonstrating the merits of sparsity and non-negativity constraints imposed on the aforementioned decompositions. Twin boundaries in GaAs nanowires were used as a test case to validate the methods, which have then been applied to a number of more complex cases.

Enhanced technique for characterization of strain distribution in crystal from EBSD patterns

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Electron backscattering diffraction method (Kikuchi method) is used for determination of crystal orientation, phase, strain and stress state of crystals. Elastic strains caused by the distortion of crystal structure change the interplanar distances [1]. The last ones reveal themselves in the changing of intensity distribution onto the Kikuchi pattern to have been measured. Due to the high spatial resolution and high locality the EBSD method can be used for studying of highly inhomogeneous samples. Weld joints of NiCrFe and artificial diamond crystals are these kinds of materials. Kikuchi patterns were obtained by using scanning electron microscope “Zeiss” Evo-50 with CCD detector.

The enhanced technique which combines two-dimensional Fourier transformation of Kikuchi pattern as whole and analysis of intensity profiles of separate Kikuchi bands was developed. The strain tensor determination was performed using the new approach which is based onto the local cross-correlation of intensity peaks of multi-beam areas.

Approbation of developed technique was carried out on two diamonds samples: one of them was obtained by temperature gradient method in Fe–Al–C system and second – in Mg–C+bor system by growth method on diamond single crystal synthesized in Ni–Mn–C system. As a result characteristic surfaces of strain tensor were constructed for local areas, which correspond to different crystalline blocks (Fig. 1).

Next, the samples of weld joint of NiCrFe nickel alloy with crack have been researched for the purpose of establishment of probable causes of crack formation. In addition to Kikuchi patterns the distribution of chemical composition was obtained by means of X-ray microanalysis. The features of strain distribution around crack were established. Particularly, it was demonstrated that maximal strain values are spreaded along the low angle boundaries of subgrains.