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The dehydrated copper silicate Na₂[Cu₂Si₄O₁₁]: a three-dimensional microporous framework with a linear Si—O—Si linkage

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Key indicators: single-crystal X-ray study; T = 298 K; mean σ (Si–O) = 0.004 Å; R factor = 0.043; wR factor = 0.120; data-to-parameter ratio = 11.9.

The structure of the title dehydrated copper silicate, disodium dicopper undecaoxide tetrasilicate, $Na_2(Cu_2O_{11}Si_4)$, was determined by single-crystal X-ray diffraction from a nonmerohedral twin. It exhibits an effective three-dimensional microporous framework with the major channels, in which the Na⁺ cations are placed, running along the *a*-axis direction and smaller channels observed along the *b*-axis direction. The structure is unusual in that it contains a symmetry-constrained Si-O-Si angle of 180°. The Cu centre is coordinated to five O atoms, exhibiting a slightly distorted square-pyramidal coordination geometry. The Na cation is interacting with five neighbouring O atoms, exhibiting an uncharacteristic coordination environment.

Related literature

For related literature, see: Brandão *et al.* (2005); Haile & Wuensch (2000); Liebau (1985); Rocha & Anderson (2000); Rocha & Lin (2005); dos Santos *et al.* (2005); Ananias *et al.* (2001, 2006); Anderson *et al.* (1994); Ferreira *et al.* (2003).



1587 measured reflections 1043 independent reflections

 $R_{\rm int}=0.042$

782 reflections with $I > 2\sigma(I)$

Experimental

Crystal data

$Na_2(Cu_2O_{11}Si_4)$	$\gamma = 100.461 \ (7)^{\circ}$
$M_r = 461.44$	V = 258.9 (2) Å ³
Triclinic, $P\overline{1}$	Z = 1
a = 5.190 (2) Å	Mo $K\alpha$ radiation
b = 6.299 (3) Å	$\mu = 4.71 \text{ mm}^{-1}$
c = 8.196 (4) Å	T = 298 (2) K
$\alpha = 96.390 \ (7)^{\circ}$	$0.28 \times 0.08 \times 0.04 \text{ mm}$
$\beta = 97.281 \ (7)^{\circ}$	

Data collection

Bruker SMART CCD 1000
diffractometer
Absorption correction: multi-scan
(TWINABS; Sheldrick, 2002)
$T_{\rm min} = 0.627, T_{\rm max} = 0.834$

Refinement

$R[F^2 > 2\sigma(F^2)] = 0.042$	88 parameters
$vR(F^2) = 0.119$	$\Delta \rho_{\rm max} = 1.40 \text{ e} \text{ Å}^{-3}$
S = 1.01	$\Delta \rho_{\rm min} = -1.58 \text{ e } \text{\AA}^{-3}$
.043 reflections	

Table 1 Selected b

iec	tea	bond	lengths	(A).	

Cu1-O5 ⁱ	1.909 (3)	Cu1-O6	1.974 (3)
Cu1-O2	1.950 (4)	Cu1-O2 ⁱⁱⁱ	2.316 (4)
Cu1-O6 ⁱⁱ	1.970 (4)		

Symmetry codes: (i) x, y - 1, z; (ii) -x + 1, -y, -z + 2; (iii) -x + 2, -y, -z + 2.

Data collection: *SMART* (Bruker, 1998); cell refinement: *SMART*; data reduction: *SAINT-Plus* (Bruker, 2003); program(s) used to solve structure: *SIR92* (Altomare *et al.*, 1993); program(s) used to refine structure: *SHELXTL* (Sheldrick, 2008); molecular graphics: *DIAMOND* (Brandenburg, 2007); software used to prepare material for publication: *SHELXTL*.

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: BR2065).

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The dehydrated copper silicate Na₂[Cu₂Si₄O₁₁]: a three-dimensional microporous framework with a linear Si—O—Si linkage

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S1. Comment

Molecular sieves containing metal cations with a range of coordination geometries have been extensively studied due to their novel topologies, interesting chemical properties and potential aplications in optoelectronics, batteries, magnetic materials and sensors (besides the traditional applications of zeolites) (Rocha & Anderson, 2000; Rocha & Lin, 2005). In the last decade, we have been interested in the synthesis and structural characterization of novel open-frameworks containing Si and metal cations (such as Ti, V, Cr, Nb, Zr and Sn) in tetrahedral and (more commonly) octahedral coordination environments, and lanthanide silicates exhibiting interesting photoluminescence properties (Anderson *et al.*, 1994; Ananias *et al.*, 2001; Ferreira *et al.*, 2003; Ananias *et al.*, 2006). As part of this research line, we prepared and characterized the hydrated copper silicate Na₂(Cu₂Si₄O₁₁).2H₂O (Brandão *et al.*, 2005). This compound was dehydrated and the magnetic properties of both hydrated and dehydrated forms were investigated (Santos *et al.*, 2005), however the crystalline structure of the dehydrated compound was not reported. Here we describe the structure of the dehydrated microporous copper silicate, Na₂(Cu₂Si₄O₁₁) (I).

The asymmetric unit of the copper silicate (I) comprises one Cu(II) cation, two corner-shared SiO₄ groups and one Na⁺ counter-cation (Figure 1). The crystallographic unique Cu(II) metal centre is coordinated to five O-atoms from five distinct SiO₄ tetrahedral moieties (four basal SiO₄ and one apical SiO₄), in a geometry resembling a distorted square pyramid for which the apical Cu—O bond is longer than the basal ones (Figure 2a and Table 1).

Adjacent SiO₄ tetrahedral moieties are linked along the *a* direction by corner-shared oxygen atoms (O3 and O4 are shared alternately) leading to the formation of zigzag metallic anionic chains, $[(Cu_2Si_4O_{11})_x]^2$, in which the Cu···Cu distances alternate between 2.9921 (8) Å (*via* bridging basal SiO₄, green bonds in Fig. 2 b) and 3.1031 (10) Å (*via* the apical SiO₄ tetrahedron, yellow bonds in Fig. 2 b). $[(Cu_2Si_4O_{11})_x]^2$ chains are interconnected *via* corner-sharing SiO₄ tetrahedra through linear interactions Si1–O1–Si^w [angle is 180.0°; symmetry code: (iv) 2 - *x*, -*y*, 1 - *z*] to form infinite layers (Fig. 2c). This linear Si–O–Si interaction is very rare and represents a remarkable structural feature of the copper silicate (I) framework. We note that such occurrence was also recently reported in the lanthanide silicate K₃(NdSi₇O₁₇) (Haile & Wuensch, 2000). From the evaluation of the structures of several hundred silicates it was concluded that the average of an unstrained Si–O–Si bond angle is *ca* 139° and that truly linear bonds are energetically unfavorable (Liebau, 1985). In fact, the crystallographically determined values of 180° are more likely to represent a time average rather than the actual value of the bond angle. The bond, at any instant in time, should have an O-atom displaced from its average position such that the instantaneous value of Si–O–Si is less than 180° (Haile & Wuensch, 2000). This structural feature is ultimately reflected in the anisotropic displacement parameters associated with this bridging O-atom. Indeed, the thermal parameters associated with this atom are unusually large, with the greatest displacement occurring in the plane perpendicular to the Si1···Si1^{iv} vector (Figure 2c).

As observed for the chains, adjacent layers are also interconnected *via* corner-sharing SiO₄ tetrahedra generating a threedimensional microporous framework with the major channels running along the *a* direction, formed by eight-membered rings and having a cross-section of *ca* 7.5×4.3 Å (Figure 3a). Interestingly, the Na⁺ cations are located within the channels but are remarkably close to the previously described layers, creating an effective porous copper framework (Figure 3a). In addition, remarkably large channels are also observed along the *b* direction, which are formed by sixmembered rings and display a cros-section of *ca* 5.2×4.6 Å (Figure 3 b).

S2. Experimental

Chemicals were purchased from commercial sources and used without further purification. An alkaline solution was prepared by mixing 13.86 g of a sodium silicate solution (Na₂O 8 wt%, SiO₂ 27 wt%), 16.13 g H₂O and 4.11 g NaOH, and a second solution was prepared by mixing 17.87 g H₂O with 7.60 g of Cu(SO₄).15H₂O. These two solutions were combined, stirred thoroughly during 2 h and the resulting gel, with a molar composition of CuO: 3.1SiO₂: 1.4Na₂O: 94.5H₂O, was autoclaved for 10 days at 503 K. A crystalline material was obtained [Na₂(Cu₂Si₄O₁₁).2H₂O], filtered and treated thermally at 573 K for six hours leads to the removal of the crystallization water molecules.

S3. Refinement

Even though crystals of the title compound could be indexed with the unit-cell parameters summarized in Table 1, a visual inspection of the centered reflections using RLATT showed the presence of a rotational twin (non-merohedral). A full sphere of reflections was collected and a partial data set was then deconvoluted using CELL_NOW (Sheldrick 2004) into a two-component twin. Data integration was performed by assuming that the second twin domain was identical to the first. The final structural model exhibits a large average U(i,j) tensor, most likely due to the applied twinning correction which ultimately seems to lead to large U3/U1 ratios.



Figure 1

Fragment of the crystal structure of the title compound with the atoms represented as thermal displacement ellipsoids drawn at the 50% probability level [Symmetry codes: (i) 2 - x, -y, 2 - z; (ii) x, -1 + y, z; (iii) 1 - x, -y, 2 - z; (iv) 2 - x, 1 - y, 2 - z; (v) x, y, 1 + z; (vi) 1 + x, y, z].



Figure 2

(*a*) Mixed ball-and-stick and polyhedral representation of the coordination environment of the Cu(II) cations and (*b*) the metallic chain $[(Cu_2Si_4O_{11})_n]^{2-}$ running along the *a* direction of the unit cell. (*c*) Schematic representation of the linear Si —O—Si bond connecting adjacent Si1 centres *via* the O1 atom. [Symmetry codes: (i) 2 - x, -y, 2 - z; (ii) x, -1 + y, z; (iii) 1 - x, -y, 2 - z; (iv) 2 - x, -y, 1 - z].



Figure 3

Perspective views of the crystal packing arrangement along the (a) [100] and (b) [010] directions of unit cell.

Z = 1

F(000) = 224 $D_x = 2.960 \text{ Mg m}^{-3}$

 $\theta = 8.1 - 58.1^{\circ}$

 $\mu = 4.71 \text{ mm}^{-1}$

 $0.28 \times 0.08 \times 0.04 \text{ mm}$

T = 298 K

Plate, black

Mo *K* α radiation, $\lambda = 0.71073$ Å

Cell parameters from 758 reflections

disodium dicopper undecaoxide tetrasilicate

Crystal data

Na₂(Cu₂O₁₁Si₄) $M_r = 461.44$ Triclinic, $P\overline{1}$ Hall symbol: -P 1 a = 5.190 (2) Å b = 6.299 (3) Å c = 8.196 (4) Å a = 96.390 (7)° $\beta = 97.281$ (7)° $\gamma = 100.461$ (7)° V = 258.9 (2) Å³

Data collection

Bruker SMART CCD 1000 diffractometer	1587 measured reflections 1043 independent reflections
Radiation source: fine-focus sealed tube	782 reflections with $I > 2\sigma(I)$
Graphite monochromator	$R_{\rm int} = 0.042$
ω scans	$\theta_{\rm max} = 26.4^\circ, \ \theta_{\rm min} = 3.9^\circ$
Absorption correction: multi-scan	$h = -6 \rightarrow 6$
(TWINABS; Sheldrick, 2002)	$k = -7 \longrightarrow 7$
$T_{\min} = 0.627, \ T_{\max} = 0.834$	$l = 0 \rightarrow 10$

Refinement

Refinement on F^2	Primary atom site location: structure-invariant
Least-squares matrix: full	direct methods
$R[F^2 > 2\sigma(F^2)] = 0.042$	Secondary atom site location: difference Fourier
$wR(F^2) = 0.119$	map
<i>S</i> = 1.01	$w = 1/[\sigma^2(F_o^2) + (0.0809P)^2]$
1043 reflections	where $P = (F_o^2 + 2F_c^2)/3$
88 parameters	$(\Delta/\sigma)_{\rm max} < 0.001$
0 restraints	$\Delta ho_{ m max} = 1.40 \ { m e} \ { m \AA}^{-3}$
	$\Delta \rho_{\rm min} = -1.58 \text{ e} \text{ Å}^{-3}$

Special details

Geometry. All e.s.d.'s (except the e.s.d. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell e.s.d.'s are taken into account individually in the estimation of e.s.d.'s in distances, angles and torsion angles; correlations between e.s.d.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell e.s.d.'s is used for estimating e.s.d.'s involving l.s. planes.

Refinement. Refinement of F^2 against ALL reflections. The weighted *R*-factor *wR* and goodness of fit *S* are based on F^2 , conventional *R*-factors *R* are based on *F*, with *F* set to zero for negative F^2 . The threshold expression of $F^2 > \sigma(F^2)$ is used only for calculating *R*-factors(gt) *etc.* and is not relevant to the choice of reflections for refinement. *R*-factors based on F^2 are statistically about twice as large as those based on *F*, and *R*- factors based on ALL data will be even larger.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters $(Å^2)$

	x	У	Ζ	$U_{ m iso}$ */ $U_{ m eq}$
Cul	0.70666 (10)	-0.10891 (9)	0.93472 (8)	0.0096 (3)

supporting information

Na1	0.8700 (4)	0.3540 (3)	1.1990 (3)	0.0235 (6)
Si1	1.0175 (2)	0.1358 (2)	0.67930 (18)	0.0087 (4)
Si2	0.5954 (3)	0.3456 (2)	0.80969 (18)	0.0089 (4)
01	1.0000	0.0000	0.5000	0.0217 (13)
O2	1.0064 (7)	-0.0190 (6)	0.8196 (5)	0.0127 (8)
03	0.7804 (6)	0.2736 (6)	0.6719 (5)	0.0126 (8)
O4	0.2923 (6)	0.3209 (5)	0.7137 (5)	0.0144 (8)
05	0.7200 (7)	0.5899 (6)	0.8873 (5)	0.0170 (9)
06	0.5974 (6)	0.1760 (5)	0.9452 (5)	0.0107 (8)

Atomic displacement parameters $(Å^2)$

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Cu1	0.0054 (4)	0.0083 (4)	0.0159 (4)	0.0011 (2)	0.0051 (2)	0.0018 (2)
Na1	0.0137 (11)	0.0161 (12)	0.0396 (16)	0.0021 (9)	0.0072 (10)	-0.0030 (10)
Si1	0.0048 (7)	0.0096 (7)	0.0124 (8)	0.0018 (5)	0.0034 (5)	0.0009 (5)
Si2	0.0040 (6)	0.0078 (7)	0.0155 (8)	0.0013 (5)	0.0042 (5)	0.0018 (5)
01	0.020 (3)	0.024 (3)	0.021 (3)	0.006 (2)	0.007 (2)	-0.006(2)
O2	0.0078 (17)	0.0137 (18)	0.018 (2)	0.0033 (13)	0.0040 (14)	0.0045 (14)
O3	0.0057 (16)	0.0169 (19)	0.018 (2)	0.0068 (14)	0.0049 (14)	0.0018 (14)
O4	0.0063 (16)	0.0140 (18)	0.023 (2)	0.0017 (14)	0.0014 (14)	0.0061 (15)
05	0.0145 (18)	0.0100 (18)	0.028 (2)	0.0000 (14)	0.0129 (16)	0.0016 (15)
06	0.0082 (16)	0.0111 (17)	0.016 (2)	0.0039 (13)	0.0061 (14)	0.0050 (14)

Geometric parameters (Å, °)

Cu1—O5 ⁱ	1.909 (3)	Si1—O4 ^{iv}	1.642 (4)
Cu1—O2	1.950 (4)	Si2—O5	1.583 (4)
Cu1—O6 ⁱⁱ	1.970 (4)	Si2—O6	1.625 (4)
Cu106	1.974 (3)	Si2—O4	1.639 (3)
Cu1—O2 ⁱⁱⁱ	2.316 (4)	Si2—O3	1.650 (4)
Cu1—Cu1 ⁱⁱ	2.9921 (13)	Si2—Cu1 ⁱⁱ	3.1221 (19)
Cu1—Cu1 ⁱⁱⁱ	3.1031 (15)	O1—Si1 ^v	1.5991 (14)
Cu1—Si2 ⁱⁱ	3.1221 (19)	O2—Cu1 ⁱⁱⁱ	2.316 (4)
Cu1—Si1	3.1673 (19)	O4—Si1 ^{vi}	1.642 (4)
Si1—O2	1.588 (4)	O5—Cu1 ^{vii}	1.909 (3)
Sil—Ol	1.5991 (14)	O6—Cu1 ⁱⁱ	1.970 (4)
Sil—O3	1.629 (3)		
O5 ⁱ —Cu1—O2	92.49 (15)	O2 ⁱⁱⁱ —Cu1—Si1	100.98 (10)
$O5^{i}$ —Cu1—O6 ⁱⁱ	91.91 (15)	Cu1 ⁱⁱ —Cu1—Si1	115.23 (4)
O2-Cu1-O6 ⁱⁱ	175.60 (13)	Cu1 ⁱⁱⁱ —Cu1—Si1	64.33 (4)
O5 ⁱ —Cu1—O6	164.27 (15)	Si2 ⁱⁱ —Cu1—Si1	179.25 (4)
O2—Cu1—O6	94.42 (14)	O2—Si1—O1	111.33 (15)
O6 ⁱⁱ —Cu1—O6	81.32 (16)	O2—Si1—O3	112.6 (2)
$O5^{i}$ —Cu1—O2 ⁱⁱⁱ	105.60 (16)	O1—Si1—O3	108.16 (15)
O2—Cu1—O2 ⁱⁱⁱ	87.05 (15)	O2-Si1-O4 ^{iv}	111.48 (19)
O6 ⁱⁱ —Cu1—O2 ⁱⁱⁱ	91.76 (14)	O1—Si1—O4 ^{iv}	108.08 (16)

O6—Cu1—O2 ⁱⁱⁱ	88.87 (14)	O3—Si1—O4 ^{iv}	104.92 (18)
O5 ⁱ —Cu1—Cu1 ⁱⁱ	131.06 (12)	O1—Si1—Cu1	115.96 (7)
O2—Cu1—Cu1 ⁱⁱ	135.03 (10)	O3—Si1—Cu1	84.08 (15)
O6 ⁱⁱ —Cu1—Cu1 ⁱⁱ	40.70 (10)	O4 ^{iv} —Si1—Cu1	129.55 (15)
O6—Cu1—Cu1 ⁱⁱ	40.62 (10)	O5—Si2—O6	113.2 (2)
O2 ⁱⁱⁱ —Cu1—Cu1 ⁱⁱ	90.41 (9)	O5—Si2—O4	111.75 (19)
O5 ⁱ —Cu1—Cu1 ⁱⁱⁱ	103.18 (12)	O6—Si2—O4	109.3 (2)
O2—Cu1—Cu1 ⁱⁱⁱ	48.19 (11)	O5—Si2—O3	107.1 (2)
O6 ⁱⁱ —Cu1—Cu1 ⁱⁱⁱ	130.50 (11)	O6—Si2—O3	107.00 (19)
O6—Cu1—Cu1 ⁱⁱⁱ	91.93 (10)	O4—Si2—O3	108.15 (19)
O2 ⁱⁱⁱ —Cu1—Cu1 ⁱⁱⁱ	38.86 (9)	O5—Si2—Cu1 ⁱⁱ	109.62 (16)
Cu1 ⁱⁱ —Cu1—Cu1 ⁱⁱⁱ	116.74 (4)	O4—Si2—Cu1 ⁱⁱ	81.61 (15)
O5 ⁱ —Cu1—Si2 ⁱⁱ	73.75 (12)	O3—Si2—Cu1 ⁱⁱ	134.72 (14)
O2—Cu1—Si2 ⁱⁱ	156.11 (11)	Si1 ^v —O1—Si1	180.0
O6 ⁱⁱ —Cu1—Si2 ⁱⁱ	26.72 (10)	Si1—O2—Cu1	126.8 (2)
O6—Cu1—Si2 ⁱⁱ	103.96 (11)	Si1—O2—Cu1 ⁱⁱⁱ	116.29 (18)
O2 ⁱⁱⁱ —Cu1—Si2 ⁱⁱ	78.31 (10)	Cu1—O2—Cu1 ⁱⁱⁱ	92.95 (15)
Cu1 ⁱⁱ —Cu1—Si2 ⁱⁱ	64.59 (4)	Si1—O3—Si2	132.7 (3)
Cu1 ⁱⁱⁱ —Cu1—Si2 ⁱⁱ	115.04 (5)	Si2—O4—Si1 ^{vi}	137.0 (2)
O5 ⁱ —Cu1—Si1	106.73 (13)	Si2—O5—Cu1 ^{vii}	152.9 (2)
O2—Cu1—Si1	23.68 (11)	Si2—O6—Cu1 ⁱⁱ	120.24 (18)
O6 ⁱⁱ —Cu1—Si1	153.39 (10)	Si2—O6—Cu1	130.3 (2)
O6—Cu1—Si1	75.74 (11)	Cu1 ⁱⁱ —O6—Cu1	98.68 (16)

Symmetry codes: (i) x, y-1, z; (ii) -x+1, -y, -z+2; (iii) -x+2, -y, -z+2; (iv) x+1, y, z; (v) -x+2, -y, -z+1; (vi) x-1, y, z; (vii) x, y+1, z.