Structural analysis of La/Al₂O₃ catalysts by La K-edge XAFS

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Structural analysis of La/Al₂O₃ catalyst materials was carried out by means of XRD and La K-edge XAFS. Influences of loading amounts and treatment temperature on the local structure around La were investigated. La atoms are supported on γ -Al₂O₃ in a highly dispersed form when a loading amount is less than 0.2 mmol g(Al₂O₃)⁻¹, and structural transformation of the La species does not take place in the temperature ranges of 873-1273 K. La species of highly loaded catalysts start to change their structure from oxidelike form to the aluminate at 1073 K. Formation of LaAlO₃ perovskite on the catalyst surface requires a concentration of La atoms above 0.5 mmol g(Al₂O₃)⁻¹ (7.5 wt% as La₂O₃).

Keywords: La₂O₃, γ-Al₂O₃, LaAlO₃, La K-edge XAFS

1. Introduction

 γ -Al₂O₃ is widely used as a support for industrial catalysts. The

phase transformation from γ - to α -phase and the decrease of the surface area, so called sintering, are serious problems in the use of the catalyst at high temperature. It is well known that addition of lanthanide (Ln) and alkaline earth element to alumina enhances its thermal stability, and there are many studies dealing with the prevention of sintering (Arai & Machida, 1996; Bettman et al., 1989; Burtin et al., 1987; Oudet et al., 1988; Schaper et al., 1983 & 1984). The surface LaAlO₃ phase formed on Al₂O₃ was proposed to enhance thermal stability of an alumina-based catalyst (Oudet et al., 1988). However, crystalline LaAlO3 was detected at a loading amount above 16%. The XRD characterization is not effective when a crystalline size was less than 100 Å. Although structural characterizations of La2O3/Al2O3 catalysts have been carried out by many research groups (Alvarez et al., 1997; Arai & Machida, 1996; Bettman et al., 1989; Capitán et al., 1995; Haack et al., 1992; Oudet et al., 1988; Schaper et al., 1983 & 1984; Shi et al., 1997), it remains unclear whether the perovskite phase forms on alumina only at a loading amount less than 10 wt% or not.

In the present study, we recorded La K-edge XAFS spectra to examine the structure of La species on alumina and their structural transformation upon thermal treatment. In the case of lanthanide, only a restricted information could be obtained from L_{III} -edge EXAFS because of the limitation of the L_{II} -edge. Especially for La,

L_{III}-edge EXAFS is available over no more than 10 Å⁻¹. Figure 1 shows La K- and L_{III}-edge EXAFS spectra of supported lanthanum oxides. As shown in the figure, multi-electron excitation from 2p4d to $(5d)^2$ was observed around 5.6 Å⁻¹ at L_{III}-edges, as in the case of La@C₈₂ (Nomura *et al.*, 1995). Therefore, characterization with Ln K-edge XAFS instead of L_{III}-edge XAFS is very important to obtain detailed structural information. The present study is the first application of lanthanide K-edge XAFS to supported catalyst materials.



Figure 1

La K- and LIII-edge EXAFS spectra of supported lanthanum oxide.

2. Experimental

La/Al₂O₃ catalysts were prepared by impregnation of γ -Al₂O₃ (JRC-ALO4; 174 m² g⁻¹) with an aqueous solution of La(NO₃)₃ (Nacalai, 99.9%) at 353 K, followed by calcination at 773 K for 5 h. These samples are referred to as x mmol La/Al₂O₃ (x stands for loading amount of La atom per gram of Al2O3). Prior to XAFS measurements, each sample was evacuated at a prescribed temperature for 0.5 h and calcined under O₂ for 1 h, followed by evacuation at the same temperature for 1 h. The X-ray absorption experiments were carried out on the BL01B1 beamline at SPring-8 (Hyogo, Japan) (Uruga et al., 1999). The ring energy was 8 GeV, and the stored current was 70-100 mA. The La K-edge X-ray absorption spectra were recorded in the transmission mode with a Si(311) two-crystal monochromator at room temperature. Data reduction was performed using a FACOM M1800 computer of Kyoto University Data Processing Center. The normalization method has been previously reported in detail (Tanaka et al., 1988). The Fourier transform was performed on EXAFS in the k range of ca. 3.2 - 14 Å⁻¹.

3. Results and Discussions

Figure 2 shows XRD patterns of La/Al₂O₃ catalyst treated at 873, 1073 and 1273 K. At a calcination temperature of 873 K, all patterns except for 3.4 mmol La/Al₂O₃ exhibited typical lines assigned to γ -Al₂O₃. On the other hand, other lines appeared on 3.4 mmol La/Al₂O₃ which were not identified with any JCPDS files. The decomposition temperature of La₂O₂CO₃ to La₂O₃ is 1020K (Squir *et al.*, 1994). Therefore, the lines were assigned to an intermediate phase between La₂O₃ and La₂O₂CO₃, which was formed at the catalyst preparation step. At a calcination temperature of 1073K, formation of LaAlO₃ perovskite was confirmed at a loading of 2.0 mmol g(Al₂O₃)⁻¹ and higher. Furthermore, it was confirmed that lanthanum species on 1.0 mmol La/Al₂O₃ transformed to LaAlO₃ at 1273K. The alumina-phase of 0.1 mmol La/Al₂O₃ and the support itself transformed to the δ-phase at 1273K.

Figure 3 shows La K-edge EXAFS spectra and their Fourier transforms (radial structure functions; RSFs) of La/Al_2O_3 catalysts treated at 873 K. The EXAFS spectra of all catalyst sample treated at 873 K, which loadings were less than 2.0 mmol g(Al_2O_3)⁻¹, were quite similar with loading and a single peak appeared on the RSF



Figure 2

Cu K α XRD patterns of La/Al₂O₃ treated at 873, 1073 and 1273 K, La₂O₃, LaAlO₃ and JCPDS files.

around 1.9 Å. The oscillation amplitude decreased monotonously with k, and little oscillation appeared in the k-region above 9 Å⁻¹, indicating that La species are supported in a highly dispersed form and that no La-(O)-La contribution exists on the catalyst sample. The remarkable degree of decay of the oscillation amplitude should be noted. The natural width (Γ) of the La K-edge (14.1 eV) is much larger than that of the LIII-edge (3.41 eV) (Krause and Oliver, 1979). The lifetime of the core hole created by absorbing an X-ray photon can be deduced from the width $h\Gamma/2\pi$ of the state. The resulting large Γ decreases the amplitude of the EXAFS oscillations (Strarns, 1984). Therefore, the amplitude of EXAFS oscillations for the La K-edge should be smaller than that for the LIII-edge. On the other hand, the EXAFS amplitude of supported Yb species at the Kedge (61.3 keV; $\Gamma = 31.9 \text{ eV}$) was stronger than that at the La Kedge (Yamamoto et al.). Then, the remarkable oscillation decay at the La K-edge was due to an extremely large apparent Debye-Waller factor, i.e., surface La species (LaO_X) have many kinds of La-O bond-length, and/or the symmetry of LaOx polyhedra on Al2O3 is quite distorted. Existence of many La-O bond lengths might cause counteracting the EXAFS oscillations. The irregular frequencies of EXAFS oscillation support this deduction.

Figure 4 shows EXAFS spectra and their RSFs of La/Al₂O₃ catalysts treated at 1073 and 1273 K and reference compounds. Crystalline La₂O₃ and LaAlO₃ perovskite exhibited strong EXAFS oscillations up to 15 Å⁻¹, and distinct peaks in their RSFs above 2Å.





La K-edge XAFS spectra and their Fourier Transforms of La/Al₂O₃ catalyst treated at 873 K.

The EXAFS spectra indicate that La species of catalysts begin to form LaAlO₃ at 1073 K when the loadings were above 1.0 mmol $g(Al_2O_3)^{-1}$. At a treatment temperature of 1273 K, formation of LaAlO₃ was confirmed on 0.5 mmol $g(Al_2O_3)^{-1}$. On the other hand, the EXAFS spectra of 0.1 and 0.2 mmol La/Al₂O₃ did not change by thermal treatment up to 1273 K. It shows that La species were supported on Al₂O₃ in a highly dispersed form without changing their local structure. Therefore, we conclude that formation of LaAlO₃ perovskite on the catalyst surface requires a certain concentration, and that La species at a loading less than 0.2 mmol $g(Al_2O_3)^{-1}$ are not affected by thermal treatment at a temperature below 1273 K.

The phase transformation of supported lanthanum species was shown in XANES spectra as well. Figure 5 shows La-K edge XANES spectra of representative catalyst samples, La2O3 and LaAlO₃. As illustrated with arrows, the energy positions of the first peak were classified to two types. A XANES spectrum of perovskite LaAlO₃ belongs to group A with a lower energy. Those of La_2O_3 and most of catalyst samples lie in group B with a higher energy although that of La₂O₃ slightly shifted to higher energy by ca. 2 eV. The positions of the first peak and the configuration for all the La/Al₂O₃ catalysts (0.1 - 3.4 mmol g(Al₂O₃)⁻¹) treated at 873 K were the same. The similarity of the XANES spectra indicates that the local structure around La species on Al2O3 resembles that of La₂O₃, rather than that of LaAlO₃. In a case of 3.4 mmol La/Al₂O₃, a sample treated at 873 K gave a similar XANES spectrum as that of La₂O₃, but those treated above 1073 K exhibited identical spectra to that of LaAlO₃. This change of XANES spectra upon thermal treatment agrees with the results of the XRD and EXAFS characterizations. It should be noted that the XANES spectrum of 0.1 mmol La/Al2O3 changed hardly by any thermal treatment, in contrast that 3.4 mmol La/Al2O3. Furthermore, 0.1 mmol La/Al2O3 gave the same XANES spectrum even after thermal treatment at 1473 K. It shows that the local structure of La species on 0.1 mmol La/Al2O3 changed little by thermal treatment, i.e., crystalline LaAlO3 never forms on La/Al₂O₃ at a low loading of 0.1 mmol $g(Al_2O_3)^{-1}$ (1.6 wt% as La₂O₃). This is not consistent with a previous ESR characterization claimed that LaAlO₃ forms on 1 mol% La/Al₂O₃ (3.1 wt% as



Figure 4

La K-edge XAFS spectra and their Fourier Transforms of La/Al₂O₃ catalyst treated at 1073 and 1273 K.

La₂O₃) calcined in air at 1273 K (Schaper *et al.*, 1983). Because the signal intensity of 1 mol% La/Al₂O₃ was remarkably weak, we assume that only a limited fraction of La species formed LaAlO₃ as ESR active species. A XANES spectrum of 0.5 mmol La/Al₂O₃ treated at 1273 K was identical to that of 0.1 mmol La/Al₂O₃. Since the formation of LaAlO₃ on the surface was confirmed in the RSF, we assume that only a part of the La species on 0.5 mmol La/Al₂O₃ transformed to the perovskite phase, and that the majority retained the original oxide-like structure after thermal treatment. We conclude that formation of crystalline LaAlO₃ on the catalyst surface requires certain concentration of La atoms, and that highly dispersed La species on Al₂O₃ hardly form LaAlO₃ perovskite.

Finally, LaAlO₃ has a perovskite structure with cubic symmetry and the first coordination sphere around La consists of twelve oxygen atoms of the same distance (Wyckoff, 1986). Nevertheless, the first peak of LaAlO₃ in the RSF at K-edge was very tiny. On the other hand, a distinct peak appears around 1.9 Å at the L_{III}-edge.



Figure 5

La K-edge XANES spectra of La/Al₂O₃, La₂O₃ and LaAlO₃. Energy was not calibrated.

Such small contribution of the light elements to EXAFS spectra was observed in the RSF of WC at the W K-edge (Borowski *et al.*, 1999). The reason why the first peak of La-O in LaAlO₃ at the K-edge was much smaller than that of La₂O₃ is unknown.

We thank Drs. H. Tanida and T. Uruga (Japan Synchrotron Radiation Research Institute, SPring-8) in carrying out the X-ray absorption experiments. The X-ray absorption experiments have been performed under the approval of the JASRI (Proposal No. 2000A0059). TY acknowledges support by JSPS Research Fellowships for Young Scientists.

References

- Alvarez, L. J., Jacobs, J. P., Sanz, F. J. & Ordiozala J. A. (1997). Solid State Ionics 95, 73-79.
- Arai, H. & Machida, M. (1996). Appl. Catal. A 138, 161-176.
- Bettman, M., Chase, R. E., Otto, K. & Weber, W. H. (1989). J. Catal. 117, 447-454.
- Borowski, M., Bowron, D. T. & de Panfilis, S. (1999). J. Synchrotron Radiat. 6, 179-181.
- Burtin, P., Brunelle, J. P., Pijiolat, M. & Soustelle, M. (1987). Appl. Catal. 34, 239.
- Capitán, M. J., Centeno, M. A., Malet, P., Carrizosa, I., Odriozola, J. A., Márquez, A. & Fernández Sanz, J. (1995). J. Phys. Chem. 99, 4655-4660.
- Haack, L. P., de Vries, J. E., Otto, K. & Chattha, M. S. (1992). Appl. Catal. A 82, 199-214.
- Krause, M. O. & Oliver, J. H. (1979). J. Phys. Chem. Ref. Data 8, 329-338.
- Nomura, M., Nakao, Y., Kikuchi, K. & Achiba, Y. (1995). Physica B 208&209, 539-540.
- Oudet, F., Courtine, P. & Vejux, A. (1988). J. Catal. 114, 112-120.
- Schaper, H., Doesburg, E. B. M. & van Reijen, L. L. (1983). Appl. Catal. 7, 211-220.
- Schaper, H., Amesz, D. L., Doesburg, E. B. M. & van Reijen, L. L. (1984). *Appl. Catal.* 9, 129-132.
- Shi, C., Walters, A.B. & Vannice, M. A. (1997). Appl. Catal. B 14, 175-188.
- Squir, G. D., Luc, H. & Puxley, D. C. (1994). Appl. Catal. A 108, 261-278.
- Strarns, D. G. (1984). Phil. Mag. B 49, 541-558
- Tanaka, T., Yamashita, H., Tsuchitani, R., Funabiki, T. & Yoshida, S. (1988). J. Chem. Soc., Faraday Trans. 184, 2987-2999.
- Uruga, T., Tanida, H., Yoneda, Y., Takeshita, K., Emura, S., Takahashi, M., Harada, M., Nishihata, Y., Kubozono, Y., Tanaka, T., Yamamoto, T., Maeda, H., Kamishima, O., Takabayashi, Y., Nakata, Y., Kimura, H., Goto, S. & Ishikawa, T. (1999). J. Synchrotron Radiat. 6, 143-145.
- Wyckoff, R. W. G. (1986). Crystal Structures, Vol. 2 (ed. 2nd), pp. 390-402. Interscience Publishers.
- Yamamoto, T., Tanaka, T., Matsuyama, T., Funabiki, T. & Yoshida, S. submitted.