PS08.01.25 CRYSTAL STRUCTURES OF $\mathrm{Hg}_{2} \mathrm{Mo}_{5} \mathrm{O}_{16}, \mathrm{HgVO}_{3}$, $\mathrm{Hg}_{2} \mathrm{VO}_{4}, \mathrm{AND} \mathrm{Hg}_{9} \mathrm{As}_{4} \mathrm{O}_{16}$. Manfred H. Möller, Annemarie L. Wessels, Wolfgang Jeitschko, Anorganisch-Chemisches Institut, Universität Münster, Wilheln-KlemmStr: 8, D-48149 Münster, Germany

The title compounds were prepared by reactions of binary and ternary oxides in evacuated silica tubes at temperatures between 400 and $550^{\circ} \mathrm{C}$. Their crystal structures were determined and refined from single-crystal X-ray diffractometer data. $\mathrm{Hg}_{2} \mathrm{Mo}_{5} \mathrm{O}_{16}: \mathrm{P} 2 / \mathrm{c}, \mathrm{a}=913.9$ (1) $\mathrm{pm}, \mathrm{b}=$ $551.3(1) \mathrm{pm}, \mathrm{c}=1429.1(2) \mathrm{pm}, \beta=110.65(1)^{\circ}, \mathrm{Z}=2, \mathrm{R}=0.043$ (1129 structure factors, 66 variable parameters). $\mathrm{Hg}_{\mathrm{V}}^{3} 3: \mathrm{P1}, \mathrm{a}=359.2(1) \mathrm{pm}$, $\mathrm{b}=475.2(1) \mathrm{pm}, \mathrm{c}=858.8(1) \mathrm{pm}, \alpha=88.32(1)^{\circ}, \beta=79.60(1)^{\circ}, \gamma=$ $89.30(1)^{\circ}, Z=2, R=0.034(1763 \mathrm{~F}, 32 \mathrm{v}) . \mathrm{Hg}_{2} \mathrm{VO}_{4}: \mathrm{P}_{1} / \mathrm{h}, \mathrm{a}=367.3(1)$ $\mathrm{pm}, \mathrm{b}=1650.3(1) \mathrm{pm}, \mathrm{c}=1425.5(1) \mathrm{pm}, \beta=89.99()^{\circ}, Z=8, \mathrm{R}=0.038$ ( $1668 \mathrm{~F}, 91 \mathrm{v}$ ). $\mathrm{Hg}_{9} \mathrm{As}_{4} \mathrm{O}_{16}: \mathrm{R} 3 \mathrm{c}, \mathrm{a}=1665.3(1) \mathrm{pm}, \mathrm{c}=1083.79(7) \mathrm{pm}$, $\mathrm{Z}=6, \mathrm{R}=0.037$ ( $1661 \mathrm{~F}, 62 \mathrm{~V}$ ).
$\mathrm{Hg}_{2} \mathrm{MO}_{5} \mathrm{O}_{16}$ and $\mathrm{HgVO}_{3}$ are $\mathrm{Hg}(\mathrm{T})$ compounds with $\mathrm{Hg}_{2}$ pairs, $\mathrm{Hg}_{2} \mathrm{VO}_{4}$ is a mixed valent $\mathrm{Hg}(\mathrm{I}, \mathrm{II})$ compound with $\mathrm{Hg}_{2}$ pairs and isolated Hg atoms. In $\mathrm{Hg}_{9} \mathrm{As}_{4} \mathrm{O}_{16}$ the mercury atoms form almost equilateral $\mathrm{Hg}_{3}$ groups with the oxidation number +4 . The molybdenum atoms have distortded octahedral coordination. These $\mathrm{MoO}_{6}$ octahedra share comers andedges, thus forming two-dimensionally infinite sheets. The vanadium atoms are irregularly surrounded by five oxygen atoms. These coordination polyhedra share edges, thus forming infinite $\left(\mathrm{VO}_{3}{ }^{1} \cdot \mathrm{n}\right.$ n chains, which are aligned parallel to each other. The tetahedral $\mathrm{AsO}_{4}$ groups are linked only via $\mathrm{Hg}_{3}$ triangles. Thus, the central atoms of the anions are in their highest oxidation state and the compounds may be represented by the formulas $\left(\mathrm{Hg}_{2}{ }^{2+}\right)_{\mathrm{n}}\left(\mathrm{MO}_{5} \mathrm{O}_{16} 6^{2-}\right)_{\mathrm{n}},\left(\mathrm{Hg}_{2^{2+}}\right)_{\mathrm{n}}\left(\mathrm{VO}_{3}{ }^{1}\right)_{2 \mathrm{n}}$, $\left(\mathrm{Hg}_{2}{ }^{2+}\right.$ 。 $2 \mathrm{HgO})_{n}\left(\mathrm{VO}_{3}{ }^{1}\right)_{2 n}$, and $\left(\mathrm{Hg}_{3}{ }^{4+}\right)_{3}\left(\mathrm{AsO}_{4}{ }^{3}\right)_{4}$.

PS08.01.26 CRYSTALSTRUCTURE OF MIXED-VALENCE $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{3}$ WITH UNUSUAL METAL DISTRIBUTION. Yoshio Okal, Takeshi Yao ${ }^{2}$ and Naoichi Yamamotol, Faculty of Integrated Human Studies ${ }^{1}$, Faculty of Engineering ${ }^{2}$, Kyoto University, Kyoto 606 Japan

In the $\mathrm{CoO}-\mathrm{VO}_{2}-\mathrm{V}_{2} \mathrm{O}_{5}$ system studied at $600^{\circ} \mathrm{C}$ only $\mathrm{CoV}_{3} \mathrm{O}_{8}$ is found as a $\mathrm{V}(\mathrm{IV}, \mathrm{V})$ mixed-valence compound ${ }^{1}$. It was originally formulated as $\mathrm{Co}_{1+4}\left(\mathrm{~V}_{3} \mathrm{O}_{8}\right)_{2}(0.90 \leq y \leq 1)$ with two polymorphs $\alpha$ and $\beta$ phases; $\alpha$ phase transforms reversibly into $\beta$ phase at 650 $\pm 8^{\circ} \mathrm{C}$. It was reported that a phase crystallizes in the body-centered orthorhombic system but the structure has remained unknown. In the present study single crystals of $\alpha-\mathrm{CoV}_{3} \mathrm{O}_{8}$ were grown in the hydrothermal $\mathrm{CoI}_{2}-\mathrm{VO}(\mathrm{OH})_{2}$ system. It crystalizes in the orthorhombic system Ibam with $\mathrm{a}=14.3298(6) \mathrm{A}, \mathrm{b}=9.8906(6) \mathrm{A}$, $\mathrm{c}=8.3950(8) \AA$ and $\mathrm{Z}=8$. The structure was refined to $\mathrm{R} / \mathrm{R}_{\mathrm{w}}=0.034 /$ 0.030 for 1558 reflections with $\mathrm{I}>3 \sigma(\mathrm{I})$. There are three kinds of metal sites, namely octahedral $\mathrm{M}(16 \mathrm{k})$ for $\mathrm{M}=\mathrm{Co}, \mathrm{V}(1)$, tetrahedral $\mathrm{V}(2)(8 \mathrm{j})$ and trigonal-bipyramidal $\mathrm{V}(3)(8 \mathrm{j})$ where M site is occupied by Co and V atoms evenly. The framework structure is constructed as that zigzag chains of edge-shared $\mathrm{MO}_{6}$ octahedra running along the $c$-axis are linked by sharing $O(5)$ atoms along the b -axis forming slabs of $\mathrm{MO}_{6}$ octahedra parallel to the bc-plane and the slabs are joined by $\mathrm{VO}_{4}$ tetrahedra and $\mathrm{VO}_{5}$ trigonal bipyramids. The valence states of metal sites were estimated as $\mathrm{Co}^{2+}$ and $\mathrm{V}^{4}+$ for M and $\mathrm{V}^{5+}$ for $\mathrm{V}(2)$ and $\mathrm{V}(3)$. It is noteworthy that the metal distribution over M site is not random but restricted to $\mathrm{Co}-\mathrm{O}(5)-\mathrm{V}$ in neighboring $\mathrm{MO}_{6}$ through $\mathrm{O}(5)$ vertex, which accounts for the even occupancies of Co and V atoms in M site and further ensures the stoichiometric composition $\mathrm{CoV}_{3} \mathrm{O}_{8}$. The magnetic susceptibility curve exhibits a sharp peak at 8 K suggesting the onset of antiferromagnetic order and the value of effective moment is well corresponding to the formula $\left(\mathrm{Co}^{2}+0.5 \mathrm{~V}^{4}+0.5\right)_{2}$ $\mathrm{V}^{5}+{ }_{2} \mathrm{O}_{8}$.

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PS08.01.27 STRUCTURE OFHIGHTEMPERATURE PHASES OFTHALLOUS NTTRATE. P. U.M. Sastry, A. Sequeira, Solid State Physics Division, B.A.R.C., Mumbai 400085, India
$\mathrm{TINO}_{3}$ exhibits structural transitions from orthorhombic phase (III) at R. T. to hexagonal phase (II) $\left(\mathrm{T}_{\mathrm{c}}=79^{\circ} \mathrm{C}\right)$ and then to cubic phase (I) ( $\mathrm{T}_{\mathrm{c}}=147^{\circ} \mathrm{C}$ ). These transitions are believed to be associated with reorientation of planar nitrate ions. The structure of phase II is reported in an earlier single crystal neutron study! In the present work, the structures of phase II (at $115^{\circ} \mathrm{C}$ ) and phase I (at $170^{\circ} \mathrm{C}$ ) are determined using powder neutron diffraction. Refinements using Rietveld technique indicate that the stuctures of phase $\mathrm{II}\left(\mathrm{Sp}, \mathrm{Gr} . \mathrm{P} 3_{1} ; \mathrm{a}=10.435(1) \AA, \mathrm{c}=7.451(1) \AA, \mathrm{Z}=9\right)$ and phase $I(S p . G r$. Pm $3 m, a=4.307(1) \AA, Z=1)$ are similar to those of $\mathrm{RbNO}_{3}$ (IV) and $\mathrm{RbNO}_{3}$ (III) respectively. The final R -values $\left(\mathrm{R}_{\mathrm{p}}\right)$ for phases II and I are $3.8 \%$ and $5.1 \%$. In all the three phases, nitrate ions are oriented perpendicular to the three edges of surrounding Tl-cubes with equal proportions in phases II and I. Transformation to phase II from phase III involves an out-of-plane flip of $90^{\circ}$ for one of the eight nitrate ions in the cell. While the nitrate ions are ordered in phases III and II, they exhibit a 12 -fold orientational disorder in phase I with one of the $\mathrm{N}-\mathrm{O}$ bonds oriented parallel to an edge of T1-cube. Refinements rule out the earlier models proposed in literature for phases II and I in which the nitrate ions are oriented normal to the diagonal of Tl-cube.

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PS08.01.28 THE STRUCTURE OF $\mathrm{La}_{26}\left(\mathrm{BO}_{3}\right)_{8} \mathrm{O}_{27}$. E. Schweda, ${ }^{2}$ K. Wurst, $b$ J.H. Lin, c M.Z. Su aInstitut für Anorganische Chemie, Universität Tübingen, Germany bInst. für Allg. und Anorg. Chemie, Universität Innsbruck, Austria cDept. of Materials Chemistry, Peking University, Beijing 100871, China

Previously $\mathrm{La}_{26}\left(\mathrm{BO}_{3}\right)_{8} \mathrm{O}_{27}$ was described as $\mathrm{La}_{3} \mathrm{BO}_{6}$ with the remark that the samples contain a small amount of unreacted oxide [1]. Very small colourless crystals of $\mathrm{La}_{26}\left(\mathrm{BO}_{3}\right)_{8} \mathrm{O}_{27}$ were grown from $\mathrm{La}_{2} \mathrm{O}_{3}$ and $\mathrm{H}_{3} \mathrm{BO}_{3}$. $\mathrm{La}_{26}\left(\mathrm{BO}_{3}\right)_{8} \mathrm{O}_{27}$ crystallizes in the monoclinic space group $\mathrm{P}_{1} / \mathrm{c}$ with the lattice constants $\mathrm{a}=692.0(1)$ $\mathrm{pm}, \mathrm{b}=1292.3(1) \mathrm{pm}, \mathrm{c}=1457.1(1) \mathrm{pm}$ and $\beta=99.41(1)^{\circ}$.

A close inspection of this structure reveals a certain relationship to the $\left(\mathrm{CaF}_{2}\right)$ fluorite structure, both from the structure and the composition point of view. At a first sight, neglecting the boron atoms in the structure, the composition will be close to " $\mathrm{LaO}_{2}$ " ( $\mathrm{La}_{26} \mathrm{O}_{52-\mathrm{x}} \mathrm{B}_{8}$ ). Furthermore the structure of $\mathrm{La}_{26}\left(\mathrm{BO}_{3}\right)_{8} \mathrm{O}_{27}$ can be considered as a distorted version of the fluorite structure. To emphasize this feature one of the "fluorite" layers with a stacking sequence O-La-O is presented in fig. 1. The sheets are oriented perpendicular to the [101] lattice direction.
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References

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