## MS15 Minerals and materials

Chairs: Frederic Hatert, Roland Nilica

# MS15-P1 Synthesis and characterization of Sb<sub>3</sub>O<sub>4</sub>F, Y<sub>0.49</sub>Sb<sub>2.51</sub>O<sub>4</sub>F, and other Sb-O-F compounds

Sk Imran Ali1, Mats Johnsson2

1. Post-doc research fellow, Stockholm University S-10691 Stockholm Sweden

2. Prof. , Department of Materials and Environmental Chemistry, Stockholm University S-10691 Stockholm Sweden

### email: skimran1984@gmail.com

Six different Sb<sup>3+</sup>-O-F compounds have previously been reported. Two are orthorhombic, designated as L-and M-SbOF,<sup>1</sup> one is cubic denoted as H-SbOF, one is glass<sup>1</sup> and the remaining two phases are monoclinic denoted as  $\alpha$ -Sb\_2O\_3F,  $\beta$ -Sb\_2O\_3F,  $^2$ -One more compound, Sb\_Q-F, is theoretically predicted from the Sb\_F-Sb\_Q-phase diagram, however, it is not yet found experimentally. All the Sb-O-F compounds show framework type of structures. The main structural unit consists of different kinds of SbQ\_3E, SbQ\_3F, SbF\_4E, SbOF\_3E, SbO\_4E polyhedral units, where Sb<sup>3+</sup> is equipped with a lone-electron-pair, E.

H-, L-, M- and the amorphous form of SbOF are synthesized by solid state reactions at different temperatures from mixtures of Sb<sub>3</sub>F and Sb<sub>2</sub>O<sub>3</sub> (1:1 ratio). The compounds  $\alpha$ -Sb<sub>2</sub>O<sub>4</sub>F<sub>5</sub> and  $\beta$ -Sb<sub>2</sub>O<sub>5</sub>F<sub>5</sub> were synthesized in an aqueous solution of NH<sub>4</sub>F and SbF<sub>3</sub> with molar ratio of 0.05:1.<sup>1</sup> Synthesis via hydrothermal techniques havenot previously not been reported for these compounds.

In this study Sb<sub>3</sub>O<sub>4</sub>F, a new Sb<sup>3+</sup>-O-F compound, has been synthesized by hydrothermal techniques. We have also synthesized  $Y_{0.5}$ Sb<sub>5</sub>O<sub>4</sub>F by introducing YF<sub>3</sub> as one of the reactants. The structural characterization is made from single crystal data will be extensively discussed. Single crystals of the two previously known compounds M-SbOF and  $\alpha$ -SbO<sub>2</sub>O<sub>5</sub>F<sub>8</sub> were also synthesized by the same technique differing from the previously known solid state synthesis. A comparison is made with previously reported compounds in the Sb3<sup>+</sup>-O-X system (X = F, Cl, Br, I).

[1] A. S. Astrom, Acta chem. Scand., 1971, 25, 1519–1520.

[2] A. A. Udovenko, L. A. Zemnukhova, E. V Kovaleva and G. A. Fedorishcheva, *Russ. J. Coord. Chem.*, 2004, 30, 618–624.

[3] A. V. Kalinchenko, F. V., Borzenkova, M.P., and Novoselova, *Zh. Neorg. Khim.*, 1983, **28**, 2426.

Keywords: Hydrothermal synthesis, Transition metal oxo-halides, Single crystal X-ray diffraction

## MS15-P2 New topology of cesium aluminum borophosphate: synthesis, crystal structure and IR-spectroscopy investigation

Vladislava I. Belik1, Larisa V. Shvanskaya1, Elena Y. Borovikova1

1. M.V. Lomonosov Moscow State University

#### email: vladislava.belik@mail.ru

Borophosphates have received much attention because of their fascinating structural architectures and potential applications in sorption, catalysis, optics and ion exchange. An open-framework cesium aluminum borophosphate, CsAl<sub>3</sub>BP<sub>6</sub>O<sub>20</sub>, with novel topology of layered anionic borophosphate partial structure was synthesized by solid state reaction method. The crystal structure was determined from single-crystal X-ray data ( $R_1 = 0.043$ ): S.G. *Pbeat*, a = 11.815(2), b = 10.042(2), c = 26.630(4) Å, Z = 8, V = 3159.5(10) Å<sup>3</sup>. The structure contains the 16-member ring borophosphate layers stacked along [001] (Fig.1) and interconnected by aluminum octahedra. The resulting three-dimensional framework is characterized by channels running parallel to [021] and [100] directions formed by six- and five-membered rings, respectively. Cs ions reside within these channels. The topological relations between the CsAl<sub>3</sub>(P,O<sub>10</sub>)<sub>2</sub> [1] and CsAl<sub>2</sub>BP<sub>6</sub>O<sub>20</sub> structures are discussed.

The MID-FTIR spectrum (Fig.1) of new borophosphate corresponds well to revealed crystal structure. Its interpretation can be made on the basis of characteristic vibrations of PO<sub>2</sub>, PO<sub>3</sub> and BO<sub>4</sub> groups, P-O-P and B-O-P bridges. The high frequency bands between 1280–1230 cm<sup>-1</sup> are attributed to the antisymmetric vibrations of O-P-O bonds v<sub>a</sub>(PO<sub>2</sub>). The bands in the region 1205 – 1150 cm<sup>-1</sup> are assigned to symmetrical vibrations of O-P-O bonds v<sub>b</sub>(PO<sub>2</sub>). The bands in the region from 1140 to 930 cm<sup>-1</sup> belong to the asymmetrical stretching vibrations of PO<sub>2</sub>, PO<sub>3</sub> and BO<sub>4</sub> units. Strong bands at 930–900 cm<sup>-1</sup> and weak bands at 775–680 cm<sup>-1</sup> can be, respectively, attributed to the antisymmetric and symmetric stretching vibrations of PO-P and B-O-P bridges. The low-frequency region (650-400 cm<sup>-1</sup>) is quite complex: bending vibrations contribute to the absorption in the 650-450 cm<sup>-1</sup> region.

The  $CsAl_2BP_6O_{20}$  presents a first example of borophosphates with anionic partial structure containing the triphosphate groups and characterizing by B:P equal to 1:6.

[1] Lesage J., Guesdon A., Raveau B. // J. Solid State Chem. 2005. V. 178. P. 1212.



Figure 1. The 2D borophosphate anionic partial structure and infrared absorption spectrum of  $CsAl_2BP_6O_{20}$ .

Keywords: New topology of borophosphate, solid state reaction method

## MS15-P3 Mixed alkali/alkaline earth trielides of the BaAl<sub>4</sub>-type structure: A combined synthetic, crystallograhic and theoretical case study for the 'coloring' in polar intermetallics

Martha Falk1, Carolin Meyer1, Matthias Kledt1, Caroline Röhr1

1. Institut für Anorganische und Analytische Chemie, Universität Freiburg

### email: martha@almandine.chemie.uni-freiburg.de

The 'coloring' [1], the distribution of different atoms *M* among the apical/basal site of the pyramids in the BaA1<sub>4</sub>-type (d), has already been extensively investigated for hundreds of ternary TM/*p*-block compounds (cf. references in [2-4]). Concerning the electronic stability the optimized 'bond energy' of 14 ve/fu is sufficiently proven [5,6], even though the structure type occurs from 12 to almost 15 ve/fu. Using metallic *M* and ionic *A*<sup>*n*+</sup> radii, the ratio  $r_M:r_A$  of the BaA1<sub>4</sub>-type ranges from 0.89 to 1.04 [4].

The 'coloring' of the *M* anion by the triels, which differ both in size and electronegativity  $\chi$ , have been systematically investigated for the Ba series (Al/Ga/In), SrGa<sub>4</sub> to SrAl<sub>4</sub> (+In, [7]) (14 ve/fu) as well as for the Ga-containing K/Rb tetraindides (13 ve/fu). Carefully performed powder/single crystal structure analysis of distinct compounds (black symbols) reveal the ThCr<sub>2</sub>Si<sub>2</sub> ordering only (*I*4/*mmm*), no indications towards the CaBe<sub>2</sub>Ge<sub>2</sub> or other 1:1:3 ordering variants are observed.

The calculated (FP-LAPW DFT) Bader volumes  $(V_{BB})$ of the binary trielides indicate no significant size differences for  $M_a$  and  $M_c$ , but a substantial more negative charge (q) of  $M_a^+$ , due to the larger Coulomb interaction  $M_a$ —A. Accordingly, all Ga-phases show a strong preference for the electronegative Ga to occupy the  $M_a$ site (red curves in (a) and (b)). The preference is more restrictive for shorter A—M contacts, i.e. smaller  $r_A$  (e.g. difference Sr/Ba in (a)). The calculated 'coloring energy' ([5],  $\Delta E_{tot}$  (CaAl<sup>b</sup>,  $Ga^2 \Leftrightarrow CaGa^b, Al^a$ , 0.46 eV) is by far larger than the difference of the  $M_a$ — $M_a$  bond energies for Al/Ga (0.14 eV).

For mixed Al/In compounds (c) the M distribution changes with  $r_{x}$ : For smaller Sr with higher 'site energy'  $A-M_a$ . In with larger  $\chi$  occupies the  $M_a$  site. In contrast, for A=Ba the less electronegative element Al occupies this site. This change of the site preference could be verified by the calculations. It is a striking example for the important contribution of Coulomb interactions in the lattice energy of polar intermetallics.

[1] G. J. Miller, Eur. J. Inorg. Chem. 523 (1998)

[2] Q. Lin et al., Z. Anorg. Allg. Chem. 641, 375 (2015)

[3] T.-S. You et al., Bull. Korean. Chem. Soc. 34, 1656 (2013)

[4] M. Wendorff, C. Röhr, Z. Naturforsch. 68b, 307 (2013)

[5] U. Häussermann et al., J. Am. Chem. Soc. **124**, 4371 (2002)

[6] M. Wendorff, C. Röhr, Z. Anorg. Allg. Chem. 631, 338 (2005)

[7] C. Meyer, K. Köhler, C. Röhr, Z. Kristallogr. Suppl. 35, 86 (2015)