

Pentaterbium lithium tristannide, Tb_5LiSn_3

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Received 30 September 2011; accepted 7 October 2011

Key indicators: single-crystal X-ray study; $T = 293$ K; mean $\sigma(\text{Tb-Sn}) = 0.001 \text{ \AA}$; R factor = 0.021; wR factor = 0.066; data-to-parameter ratio = 15.4.

The new ternary phase pentaterbium lithium tristannide, Tb_5LiSn_3 , crystallizes in the hexagonal Hf_5CuSn_3 structure type, which is a ‘filled’ version of the binary $RE_5\text{Sn}_3$ phases (Mn_5Si_3 -type) (RE is rare earth). The asymmetric unit contains two Tb sites (site symmetries 3.2 and $m2m$), one Li site (site symmetry $\bar{3}m$) and one Sn site (site symmetry $m2m$). The 14-vertex Frank–Kasper polyhedra are typical for Li and Tb atoms. The environment of the Sn atom is a pseudo-Frank–Kasper polyhedron with a coordination number of 13 for the tin atom. One of the Tb atoms is enclosed in a 17-vertex polyhedron. The metallic type of bonding was indicated by an analysis of the interatomic distances.

Related literature

For the Hf_5CuSn_3 structure type, see: Rieger & Parthé (1965). For related structures, see: Pavlyuk & Bodak (1992a,b); Pavlyuk *et al.* (1989, 1991, 1993). For the magnetic properties of related compounds, see: Tran *et al.* (2008).

Experimental

Crystal data

Tb_5LiSn_3

$M_r = 1157.72$

Hexagonal, $P6_3/mcm$
 $a = 9.0122 (14) \text{ \AA}$
 $c = 6.5744 (13) \text{ \AA}$
 $V = 462.4 (2) \text{ \AA}^3$
 $Z = 2$

Mo $K\alpha$ radiation
 $\mu = 45.56 \text{ mm}^{-1}$
 $T = 293 \text{ K}$
 $0.07 \times 0.05 \times 0.03 \text{ mm}$

Data collection

Oxford Diffraction Xcalibur3 CCD diffractometer
Absorption correction: analytical (*CrysAlis RED*; Oxford Diffraction, 2008)
 $T_{\min} = 0.322$, $T_{\max} = 0.657$

1907 measured reflections
216 independent reflections
207 reflections with $I > 2\sigma(I)$
 $R_{\text{int}} = 0.021$

Refinement

$R[F^2 > 2\sigma(F^2)] = 0.021$
 $wR(F^2) = 0.066$
 $S = 1.33$
216 reflections

14 parameters
 $\Delta\rho_{\max} = 1.08 \text{ e \AA}^{-3}$
 $\Delta\rho_{\min} = -1.47 \text{ e \AA}^{-3}$

Data collection: *CrysAlis CCD* (Oxford Diffraction, 2008); cell refinement: *CrysAlis CCD*; data reduction: *CrysAlis RED* (Oxford Diffraction, 2008); program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: *DIAMOND* (Brandenburg, 2006); software used to prepare material for publication: *SHELXL97*.

Financial support from the Ministry of Education and Science, Youth and Sport of Ukraine (N 0111U001089) is gratefully acknowledged.

Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: FF2031).

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supporting information

Acta Cryst. (2011). E67, i61 [doi:10.1107/S1600536811041328]

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S1. Comment

The RE_5TM_3 (RE - rare earth, T - Cu , Ag and M - Sn , Pb) ternary stannides crystallize in a hexagonal Hf_5CuSn_3 (superstructure to Ti_5Ga_4 -type) with space group $P6_3/mcm$ (Rieger and Parthé, 1965). These intermetallic compounds are characterized by two different sites for the RE atoms located at $4\ d$ and $6\ g$, respectively. The Sn or Pb atoms are located at the next $6\ g$ site and the transition atoms occupy $2\ b$ site. The RE_5TM_3 intermetallics are 'filled' version of the binary RE_5M_3 phases which crystallize in Mn_5Si_3 structure type. It is also possible, that the transition metals fill the octahedral voids.

For the Ce based compounds, Ce_5TM_3 , investigated by (Tran *et al.*, 2008) are found multiple magnetic phase transitions at low temperatures and discussed the role of f - spd hybridization on the evolution of heavy-fermion behaviour.

We detected the new ternary compound during the systematic study of ternary alloys of $\text{Tb}-\text{Li}-\text{Sn}$ system from the concentration region with low content of lithium. The powder diffraction pattern of this compound is similar to the powder pattern of the RE_5Sn_3 (RE - rare-earth metals) binary phases, but has some differences. So we decided to further study this phase using single-crystal method. Obtained single-crystal data show that the title compound crystallizes with the hexagonal space group $P6_3/mcm$ as a Hf_5CuSn_3 type. The projection of the unit cell and coordination polyhedra of the atoms are shown in Fig. 1. The distribution of tin and lithium atoms in three-dimensional-nets consisted of Tb atoms are shown in Fig. 2.

The number of neighbouring atoms correlates well with the dimensions of the central atoms. The Tb atoms are enclosed in 14- and 17-vertex polyhedra. The coordination polyhedron of the Sn atom is pseudo Frank-Kasper polyhedron with $\text{CN}=13$. Lithium atom is surrounded by 14 neighbours atoms in the form of 14-vertex Frank-Kasper polyhedron. The shortest interatomic distances in the title compound are in the typical for intermetallic compounds ranges and indicate metallic type of bonding.

In the title compound lithium atoms occupy the same crystallographic position that the atoms of transition metal in the original structure type. The same was observed previously when we studied RELiSn_2 compounds with the CeNiSi_2 structure type (Pavlyuk *et al.*, 1989), RELiGe with the ZrNiAl type (Pavlyuk *et al.*, 1991 and Pavlyuk & Bodak, 1992a), $\text{RE}_3\text{Li}_2\text{Ge}_3$ with $\text{Hf}_3\text{Ni}_2\text{Si}_3$ type (Pavlyuk & Bodak, 1992b), solid solutions $\text{RLi}_x\text{Cu}_{2-x}\text{Si}_2$ and $\text{RLi}_x\text{Cu}_{2-x}\text{Ge}_2$ (Pavlyuk *et al.*, 1993).

S2. Experimental

Terbium, lithium and tin, all with a nominal purity more than 99.9 wt. %, were used as starting elements. First, the pieces of the pure metals with a stoichiometry $\text{Tb}_{55}\text{Li}_{10}\text{Sn}_{35}$ were pressed into pellet, enclosed in tantalum crucible and placed in a resistance furnace with a thermocouple controller. Heating rate from room temperature to 670 K was equal 5 K per minute. At this temperature the alloy was held over 2 d and then the temperature was increased from 670 to 1070 K over 1 h. Then the alloy was annealed at this temperature for 8 h and slowly cooled down to room temperature. After the

melting and annealing procedures, the total weight loss was less than 2%. Small good quality single-crystal of the title compound was isolated from alloy.

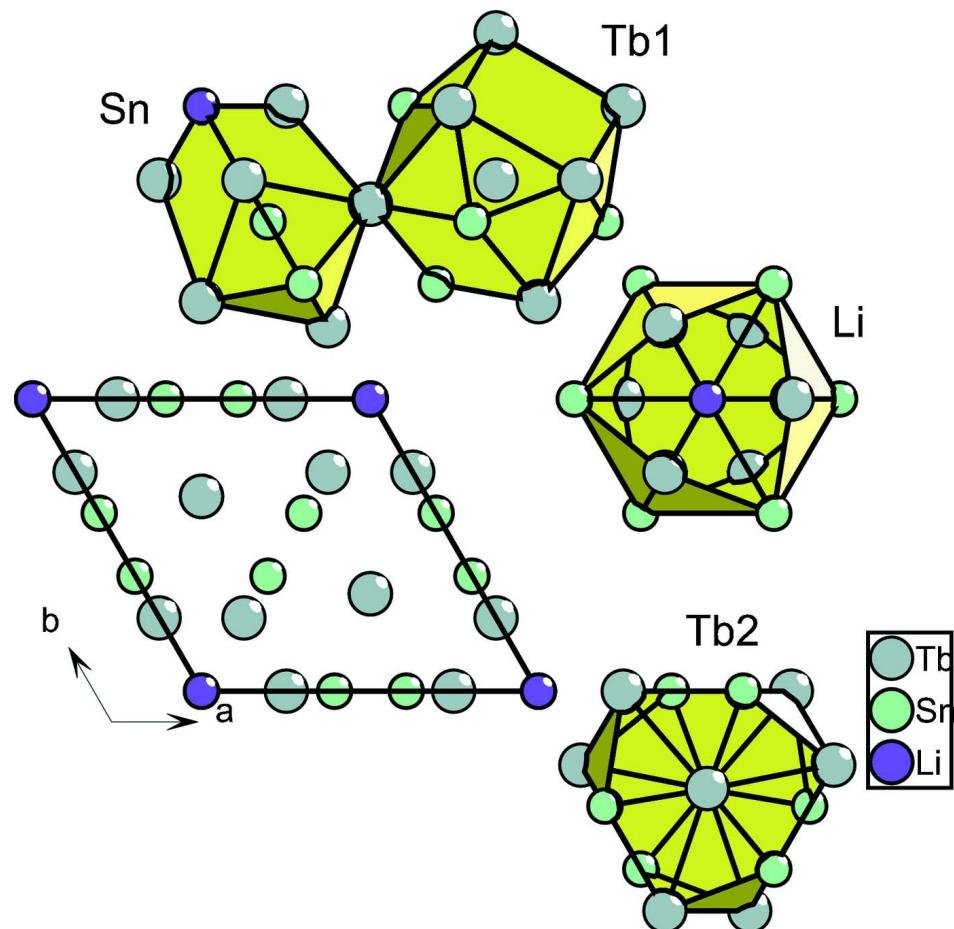
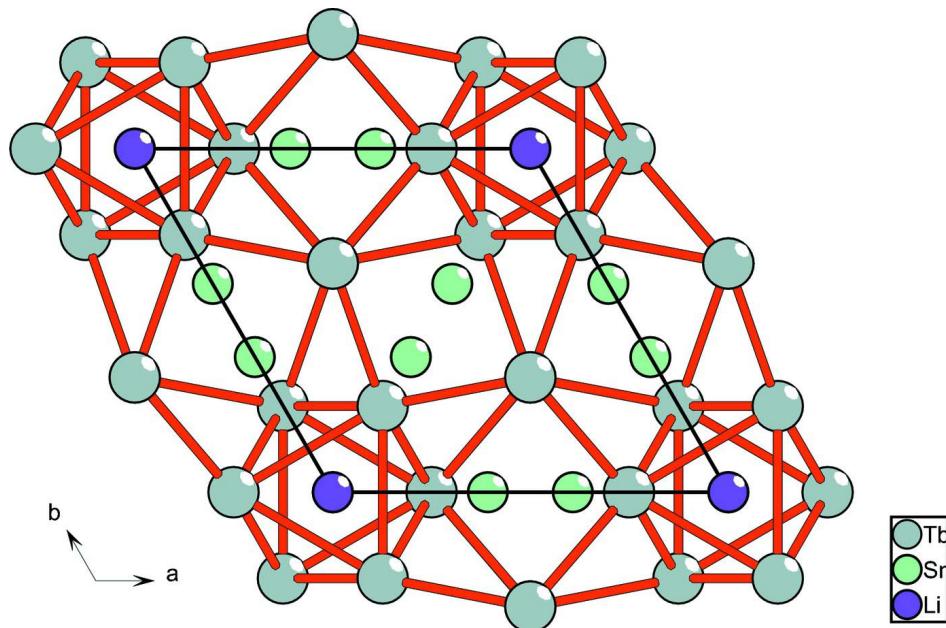


Figure 1

The projection of the unit cell and coordination polyhedra of the atoms.

**Figure 2**

The distribution of tin and lithium atoms in three-dimensional-nets consisted of Tb atoms.

Pentaterbium lithium tristannide

Crystal data

Tb_5LiSn_3
 $M_r = 1157.72$
 Hexagonal, $P6_3/mcm$
 Hall symbol: -P 6c 2
 $a = 9.0122 (14)$ Å
 $c = 6.5744 (13)$ Å
 $V = 462.4 (2)$ Å³
 $Z = 2$
 $F(000) = 956$

$D_x = 8.315 \text{ Mg m}^{-3}$
 Mo $K\alpha$ radiation, $\lambda = 0.71073$ Å
 Cell parameters from 1907 reflections
 $\theta = 2.6\text{--}27.4^\circ$
 $\mu = 45.56 \text{ mm}^{-1}$
 $T = 293$ K
 Prism, metallic dark grey
 $0.07 \times 0.05 \times 0.03$ mm

Data collection

Oxford Diffraction Xcalibur3 CCD
 diffractometer
 Radiation source: fine-focus sealed tube
 Graphite monochromator
 Detector resolution: 0 pixels mm⁻¹
 ω scans
 Absorption correction: analytical
 (*CrysAlis RED*; Oxford Diffraction, 2008)
 $T_{\min} = 0.322$, $T_{\max} = 0.657$

1907 measured reflections
 216 independent reflections
 207 reflections with $I > 2\sigma(I)$
 $R_{\text{int}} = 0.021$
 $\theta_{\max} = 27.4^\circ$, $\theta_{\min} = 2.6^\circ$
 $h = -11 \rightarrow 11$
 $k = -11 \rightarrow 11$
 $l = 0 \rightarrow 8$

Refinement

Refinement on F^2
 Least-squares matrix: full
 $R[F^2 > 2\sigma(F^2)] = 0.021$
 $wR(F^2) = 0.066$
 $S = 1.33$
 216 reflections

14 parameters
 0 restraints
 Primary atom site location: structure-invariant
 direct methods
 Secondary atom site location: difference Fourier
 map

$$w = 1/[\sigma^2(F_o^2) + (0.0268P)^2 + 3.977P]$$

where $P = (F_o^2 + 2F_c^2)/3$
 $(\Delta/\sigma)_{\max} < 0.001$

$$\Delta\rho_{\max} = 1.08 \text{ e } \text{\AA}^{-3}$$

$$\Delta\rho_{\min} = -1.47 \text{ e } \text{\AA}^{-3}$$

Special details

Geometry. All e.s.d.'s (except the e.s.d. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell e.s.d.'s are taken into account individually in the estimation of e.s.d.'s in distances, angles and torsion angles; correlations between e.s.d.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell e.s.d.'s is used for estimating e.s.d.'s involving l.s. planes.

Refinement. Refinement of F^2 against ALL reflections. The weighted R -factor wR and goodness of fit S are based on F^2 , conventional R -factors R are based on F , with F set to zero for negative F^2 . The threshold expression of $F^2 > \sigma(F^2)$ is used only for calculating R -factors(gt) etc. and is not relevant to the choice of reflections for refinement. R -factors based on F^2 are statistically about twice as large as those based on F , and R -factors based on ALL data will be even larger.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters (\AA^2)

	x	y	z	$U_{\text{iso}}^*/U_{\text{eq}}$
Tb1	0.25088 (10)	0.0000	0.2500	0.0479 (3)
Tb2	0.3333	0.6667	0.0000	0.0535 (3)
Sn3	0.60694 (14)	0.0000	0.2500	0.0493 (4)
Li4	0.0000	0.0000	0.0000	0.055 (13)

Atomic displacement parameters (\AA^2)

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Tb1	0.0478 (4)	0.0481 (5)	0.0479 (5)	0.0241 (3)	0.000	0.000
Tb2	0.0536 (4)	0.0536 (4)	0.0532 (6)	0.0268 (2)	0.000	0.000
Sn3	0.0491 (5)	0.0493 (7)	0.0496 (6)	0.0247 (4)	0.000	0.000
Li4	0.07 (2)	0.07 (2)	0.03 (2)	0.033 (11)	0.000	0.000

Geometric parameters (\AA , $^\circ$)

Tb1—Li4	2.7952 (8)	Tb2—Tb1 ^{xii}	3.8093 (7)
Tb1—Li4 ⁱ	2.7952 (8)	Tb2—Tb1 ^{xi}	3.8093 (7)
Tb1—Sn3 ⁱⁱ	3.1066 (11)	Tb2—Tb1 ^{xiii}	3.8093 (8)
Tb1—Sn3 ⁱⁱⁱ	3.1066 (11)	Tb2—Tb1 ^{ix}	3.8093 (7)
Tb1—Sn3	3.2090 (16)	Sn3—Tb1 ^{xvii}	3.1066 (11)
Tb1—Sn3 ^{iv}	3.5281 (8)	Sn3—Tb1 ^{xviii}	3.1066 (11)
Tb1—Sn3 ^v	3.5281 (8)	Sn3—Tb2 ^{vii}	3.2247 (6)
Tb1—Tb2 ^{vi}	3.8093 (7)	Sn3—Tb2 ^{vi}	3.2247 (6)
Tb1—Tb2 ^{vii}	3.8093 (7)	Sn3—Tb2 ^{ix}	3.2247 (6)
Tb1—Tb2 ^{viii}	3.8093 (7)	Sn3—Tb2 ^{viii}	3.2247 (6)
Tb1—Tb2 ^{ix}	3.8093 (7)	Sn3—Tb1 ^{iv}	3.5281 (8)
Tb1—Tb1 ^x	3.9161 (17)	Sn3—Tb1 ^v	3.5281 (8)
Tb2—Sn3 ^{xi}	3.2247 (6)	Li4—Tb1 ^{xix}	2.7952 (8)
Tb2—Sn3 ^{xii}	3.2247 (6)	Li4—Tb1 ^{xx}	2.7952 (8)
Tb2—Sn3 ^{xiii}	3.2247 (6)	Li4—Tb1 ^{xiii}	2.7952 (8)
Tb2—Sn3 ^{ix}	3.2247 (6)	Li4—Tb1 ^x	2.7952 (8)
Tb2—Sn3 ^{xiv}	3.2247 (6)	Li4—Tb1 ^{xiv}	2.7952 (8)

Tb2—Sn3 ⁱⁱⁱ	3.2247 (6)	Li4—Li4 ⁱ	3.2872 (6)
Tb2—Tb2 ^{xv}	3.2872 (6)	Li4—Li4 ^{xxi}	3.2872 (6)
Tb2—Tb2 ^{xvi}	3.2872 (6)		
Li4—Tb1—Li4 ⁱ	72.03 (3)	Sn3 ^{xiv} —Tb2—Tb1 ^{xii}	144.94 (2)
Li4—Tb1—Sn3 ⁱⁱ	82.67 (2)	Sn3 ⁱⁱⁱ —Tb2—Tb1 ^{xii}	123.785 (13)
Li4 ⁱ —Tb1—Sn3 ⁱⁱ	82.67 (2)	Tb2 ^{xv} —Tb2—Tb1 ^{xii}	64.439 (7)
Li4—Tb1—Sn3 ⁱⁱⁱ	82.67 (2)	Tb2 ^{xvi} —Tb2—Tb1 ^{xii}	115.561 (7)
Li4 ⁱ —Tb1—Sn3 ⁱⁱⁱ	82.67 (2)	Sn3 ^{xi} —Tb2—Tb1 ^{xi}	53.50 (2)
Sn3 ⁱⁱ —Tb1—Sn3 ⁱⁱⁱ	161.86 (5)	Sn3 ^{xii} —Tb2—Tb1 ^{xi}	110.39 (2)
Li4—Tb1—Sn3	143.985 (13)	Sn3 ^{xiii} —Tb2—Tb1 ^{xi}	59.520 (16)
Li4 ⁱ —Tb1—Sn3	143.985 (13)	Sn3 ^{ix} —Tb2—Tb1 ^{xi}	51.60 (2)
Sn3 ⁱⁱ —Tb1—Sn3	99.07 (2)	Sn3 ^{xiv} —Tb2—Tb1 ^{xi}	123.785 (13)
Sn3 ⁱⁱⁱ —Tb1—Sn3	99.07 (2)	Sn3 ⁱⁱⁱ —Tb2—Tb1 ^{xi}	144.94 (2)
Li4—Tb1—Sn3 ^{iv}	147.31 (3)	Tb2 ^{xv} —Tb2—Tb1 ^{xi}	115.561 (7)
Li4 ⁱ —Tb1—Sn3 ^{iv}	75.28 (2)	Tb2 ^{xvi} —Tb2—Tb1 ^{xi}	64.439 (7)
Sn3 ⁱⁱ —Tb1—Sn3 ^{iv}	93.283 (5)	Tb1 ^{xii} —Tb2—Tb1 ^{xi}	63.16 (2)
Sn3 ⁱⁱⁱ —Tb1—Sn3 ^{iv}	93.283 (5)	Sn3 ^{xi} —Tb2—Tb1 ^{xiii}	59.520 (16)
Sn3—Tb1—Sn3 ^{iv}	68.70 (3)	Sn3 ^{xii} —Tb2—Tb1 ^{xiii}	123.785 (13)
Li4—Tb1—Sn3 ^v	75.28 (2)	Sn3 ^{xiii} —Tb2—Tb1 ^{xiii}	53.50 (2)
Li4 ⁱ —Tb1—Sn3 ^v	147.31 (3)	Sn3 ^{ix} —Tb2—Tb1 ^{xiii}	144.94 (2)
Sn3 ⁱⁱ —Tb1—Sn3 ^v	93.283 (5)	Sn3 ^{xiv} —Tb2—Tb1 ^{xiii}	110.39 (2)
Sn3 ⁱⁱⁱ —Tb1—Sn3 ^v	93.283 (5)	Sn3 ⁱⁱⁱ —Tb2—Tb1 ^{xiii}	51.60 (2)
Sn3—Tb1—Sn3 ^v	68.70 (3)	Tb2 ^{xv} —Tb2—Tb1 ^{xiii}	64.439 (7)
Sn3 ^{iv} —Tb1—Sn3 ^v	137.41 (5)	Tb2 ^{xvi} —Tb2—Tb1 ^{xiii}	115.561 (7)
Li4—Tb1—Tb2 ^{vi}	102.887 (9)	Tb1 ^{xii} —Tb2—Tb1 ^{xiii}	102.753 (8)
Li4 ⁱ —Tb1—Tb2 ^{vi}	136.924 (8)	Tb1 ^{xi} —Tb2—Tb1 ^{xiii}	93.848 (16)
Sn3 ⁱⁱ —Tb1—Tb2 ^{vi}	54.444 (14)	Sn3 ^{xi} —Tb2—Tb1 ^{ix}	123.785 (13)
Sn3 ⁱⁱⁱ —Tb1—Tb2 ^{vi}	140.12 (2)	Sn3 ^{xii} —Tb2—Tb1 ^{ix}	59.520 (16)
Sn3—Tb1—Tb2 ^{vi}	53.886 (12)	Sn3 ^{xiii} —Tb2—Tb1 ^{ix}	144.94 (2)
Sn3 ^{iv} —Tb1—Tb2 ^{vi}	100.83 (2)	Sn3 ^{ix} —Tb2—Tb1 ^{ix}	53.50 (2)
Sn3 ^v —Tb1—Tb2 ^{vi}	51.970 (13)	Sn3 ^{xiv} —Tb2—Tb1 ^{ix}	51.60 (2)
Li4—Tb1—Tb2 ^{vii}	136.924 (8)	Sn3 ⁱⁱⁱ —Tb2—Tb1 ^{ix}	110.39 (2)
Li4 ⁱ —Tb1—Tb2 ^{vii}	102.887 (9)	Tb2 ^{xv} —Tb2—Tb1 ^{ix}	115.561 (7)
Sn3 ⁱⁱ —Tb1—Tb2 ^{vii}	140.12 (2)	Tb2 ^{xvi} —Tb2—Tb1 ^{ix}	64.439 (7)
Sn3 ⁱⁱⁱ —Tb1—Tb2 ^{vii}	54.444 (14)	Tb1 ^{xii} —Tb2—Tb1 ^{ix}	93.848 (16)
Sn3—Tb1—Tb2 ^{vii}	53.886 (12)	Tb1 ^{xi} —Tb2—Tb1 ^{ix}	102.753 (8)
Sn3 ^{iv} —Tb1—Tb2 ^{vii}	51.970 (13)	Tb1 ^{xiii} —Tb2—Tb1 ^{ix}	160.55 (2)
Sn3 ^v —Tb1—Tb2 ^{vii}	100.83 (2)	Tb1 ^{xvii} —Sn3—Tb1 ^{xviii}	78.14 (5)
Tb2 ^{vi} —Tb1—Tb2 ^{vii}	107.77 (2)	Tb1 ^{xvii} —Sn3—Tb1	140.93 (2)
Li4—Tb1—Tb2 ^{viii}	136.924 (8)	Tb1 ^{xviii} —Sn3—Tb1	140.93 (2)
Li4 ⁱ —Tb1—Tb2 ^{viii}	102.887 (9)	Tb1 ^{xvii} —Sn3—Tb2 ^{vii}	73.951 (16)
Sn3 ⁱⁱ —Tb1—Tb2 ^{viii}	54.444 (14)	Tb1 ^{xviii} —Sn3—Tb2 ^{vii}	137.78 (3)
Sn3 ⁱⁱⁱ —Tb1—Tb2 ^{viii}	140.12 (2)	Tb1—Sn3—Tb2 ^{vii}	72.61 (2)
Sn3—Tb1—Tb2 ^{viii}	53.886 (12)	Tb1 ^{xvii} —Sn3—Tb2 ^{vi}	137.78 (3)
Sn3 ^{iv} —Tb1—Tb2 ^{viii}	51.970 (13)	Tb1 ^{xviii} —Sn3—Tb2 ^{vi}	73.951 (16)
Sn3 ^v —Tb1—Tb2 ^{viii}	100.83 (2)	Tb1—Sn3—Tb2 ^{vi}	72.61 (2)
Tb2 ^{vi} —Tb1—Tb2 ^{viii}	51.122 (14)	Tb2 ^{vii} —Sn3—Tb2 ^{vi}	145.22 (4)

Tb2 ^{vii} —Tb1—Tb2 ^{viii}	86.152 (16)	Tb1 ^{xvii} —Sn3—Tb2 ^{ix}	73.951 (16)
Li4—Tb1—Tb2 ^{ix}	102.887 (9)	Tb1 ^{xviii} —Sn3—Tb2 ^{ix}	137.78 (3)
Li4 ⁱ —Tb1—Tb2 ^{ix}	136.924 (8)	Tb1—Sn3—Tb2 ^{ix}	72.61 (2)
Sn3 ⁱⁱ —Tb1—Tb2 ^{ix}	140.12 (2)	Tb2 ^{vii} —Sn3—Tb2 ^{ix}	61.287 (15)
Sn3 ⁱⁱⁱ —Tb1—Tb2 ^{ix}	54.444 (14)	Tb2 ^{vi} —Sn3—Tb2 ^{ix}	107.56 (2)
Sn3—Tb1—Tb2 ^{ix}	53.886 (12)	Tb1 ^{xvii} —Sn3—Tb2 ^{viii}	137.78 (3)
Sn3 ^{iv} —Tb1—Tb2 ^{ix}	100.83 (2)	Tb1 ^{xviii} —Sn3—Tb2 ^{viii}	73.951 (16)
Sn3 ^v —Tb1—Tb2 ^{ix}	51.970 (13)	Tb1—Sn3—Tb2 ^{viii}	72.61 (2)
Tb2 ^{vi} —Tb1—Tb2 ^{ix}	86.152 (16)	Tb2 ^{vii} —Sn3—Tb2 ^{viii}	107.56 (2)
Tb2 ^{vii} —Tb1—Tb2 ^{ix}	51.122 (14)	Tb2 ^{vi} —Sn3—Tb2 ^{viii}	61.287 (15)
Tb2 ^{viii} —Tb1—Tb2 ^{ix}	107.77 (2)	Tb2 ^{ix} —Sn3—Tb2 ^{viii}	145.22 (4)
Li4—Tb1—Tb1 ^x	45.533 (9)	Tb1 ^{xvii} —Sn3—Tb1 ^{iv}	73.62 (2)
Li4 ⁱ —Tb1—Tb1 ^x	45.533 (9)	Tb1 ^{xviii} —Sn3—Tb1 ^{iv}	73.62 (2)
Sn3 ⁱⁱ —Tb1—Tb1 ^x	50.93 (2)	Tb1—Sn3—Tb1 ^{iv}	111.30 (3)
Sn3 ⁱⁱⁱ —Tb1—Tb1 ^x	110.93 (2)	Tb2 ^{vii} —Sn3—Tb1 ^{iv}	68.510 (11)
Sn3—Tb1—Tb1 ^x	150.0	Tb2 ^{vi} —Sn3—Tb1 ^{iv}	125.693 (6)
Sn3 ^{iv} —Tb1—Tb1 ^x	108.33 (2)	Tb2 ^{ix} —Sn3—Tb1 ^{iv}	125.693 (6)
Sn3 ^v —Tb1—Tb1 ^x	108.33 (2)	Tb2 ^{viii} —Sn3—Tb1 ^{iv}	68.510 (11)
Tb2 ^{vi} —Tb1—Tb1 ^x	99.726 (11)	Tb1 ^{xvii} —Sn3—Tb1 ^v	73.62 (2)
Tb2 ^{vii} —Tb1—Tb1 ^x	148.420 (11)	Tb1 ^{xviii} —Sn3—Tb1 ^v	73.62 (2)
Tb2 ^{viii} —Tb1—Tb1 ^x	99.726 (11)	Tb1—Sn3—Tb1 ^v	111.30 (3)
Tb2 ^{ix} —Tb1—Tb1 ^x	148.420 (11)	Tb2 ^{vii} —Sn3—Tb1 ^v	125.693 (6)
Sn3 ^{xi} —Tb2—Sn3 ^{xii}	163.38 (4)	Tb2 ^{vi} —Sn3—Tb1 ^v	68.510 (11)
Sn3 ^{xi} —Tb2—Sn3 ^{xiii}	72.44 (2)	Tb2 ^{ix} —Sn3—Tb1 ^v	68.510 (11)
Sn3 ^{xii} —Tb2—Sn3 ^{xiii}	96.334 (10)	Tb2 ^{viii} —Sn3—Tb1 ^v	125.693 (6)
Sn3 ^{xi} —Tb2—Sn3 ^{ix}	96.334 (10)	Tb1 ^{iv} —Sn3—Tb1 ^v	137.41 (5)
Sn3 ^{xii} —Tb2—Sn3 ^{ix}	72.44 (2)	Tb1 ^{xix} —Li4—Tb1 ^{xx}	88.933 (19)
Sn3 ^{xiii} —Tb2—Sn3 ^{ix}	97.06 (4)	Tb1 ^{xix} —Li4—Tb1	91.067 (19)
Sn3 ^{xi} —Tb2—Sn3 ^{xiv}	96.334 (10)	Tb1 ^{xx} —Li4—Tb1	180.0
Sn3 ^{xii} —Tb2—Sn3 ^{xiv}	97.06 (4)	Tb1 ^{xix} —Li4—Tb1 ^{xiii}	180.0
Sn3 ^{xi} —Tb2—Sn3 ⁱⁱⁱ	163.38 (4)	Tb1 ^{xx} —Li4—Tb1 ^{xiii}	91.067 (19)
Sn3 ^{xii} —Tb2—Sn3 ⁱⁱⁱ	96.334 (10)	Tb1—Li4—Tb1 ^{xiii}	88.933 (19)
Sn3 ^{xiii} —Tb2—Sn3 ⁱⁱⁱ	97.06 (4)	Tb1 ^{xix} —Li4—Tb1 ^x	91.067 (19)
Sn3 ^{xi} —Tb2—Tb2 ^{xx}	120.643 (7)	Tb1 ^{xx} —Li4—Tb1 ^x	91.067 (19)
Sn3 ^{xii} —Tb2—Tb2 ^{xx}	59.357 (8)	Tb1—Li4—Tb1 ^{xiv}	88.933 (19)
Sn3 ^{xiii} —Tb2—Tb2 ^{xx}	59.357 (7)	Tb1 ^{xiii} —Li4—Tb1 ^{xiv}	91.067 (19)
Sn3 ^{ix} —Tb2—Tb2 ^{xx}	120.643 (8)	Tb1 ^x —Li4—Tb1 ^{xiv}	180.00 (7)
Sn3 ^{xiv} —Tb2—Tb2 ^{xx}	120.643 (7)	Tb1 ^{xix} —Li4—Li4 ⁱ	126.015 (13)
Sn3 ⁱⁱⁱ —Tb2—Tb2 ^{xx}	59.357 (7)	Tb1 ^{xx} —Li4—Li4 ⁱ	126.015 (13)
Sn3 ^{xi} —Tb2—Tb2 ^{xvi}	59.357 (7)	Tb1—Li4—Li4 ⁱ	53.985 (13)
Sn3 ^{xii} —Tb2—Tb2 ^{xvi}	120.643 (7)	Tb1 ^{xiii} —Li4—Li4 ⁱ	53.985 (13)
Sn3 ^{xiii} —Tb2—Tb2 ^{xvi}	120.643 (8)	Tb1 ^x —Li4—Li4 ⁱ	53.985 (13)
Sn3 ^{ix} —Tb2—Tb2 ^{xvi}	59.357 (8)	Tb1 ^{xiv} —Li4—Li4 ⁱ	126.015 (13)
Sn3 ^{xiv} —Tb2—Tb2 ^{xvi}	59.357 (7)	Tb1 ^{xix} —Li4—Li4 ^{xxi}	53.985 (13)

$\text{Sn3}^{\text{iii}}\text{—Tb2—Tb2}^{\text{xvi}}$	120.643 (7)	$\text{Tb1}^{\text{xx}}\text{—Li4—Li4}^{\text{xxi}}$	53.985 (13)
$\text{Tb2}^{\text{xv}}\text{—Tb2—Tb2}^{\text{xvi}}$	180.0	$\text{Tb1—Li4—Li4}^{\text{xxi}}$	126.015 (13)
$\text{Sn3}^{\text{xii}}\text{—Tb2—Tb1}^{\text{xii}}$	110.39 (2)	$\text{Tb1}^{\text{xiii}}\text{—Li4—Li4}^{\text{xxi}}$	126.015 (13)
$\text{Sn3}^{\text{xii}}\text{—Tb2—Tb1}^{\text{xii}}$	53.50 (2)	$\text{Tb1}^{\text{x}}\text{—Li4—Li4}^{\text{xxi}}$	126.015 (13)
$\text{Sn3}^{\text{xiii}}\text{—Tb2—Tb1}^{\text{xii}}$	51.60 (2)	$\text{Tb1}^{\text{xiv}}\text{—Li4—Li4}^{\text{xxi}}$	53.985 (13)
$\text{Sn3}^{\text{ix}}\text{—Tb2—Tb1}^{\text{xii}}$	59.520 (16)	$\text{Li4}^{\text{i}}\text{—Li4—Li4}^{\text{xxi}}$	180.0

Symmetry codes: (i) $-x, -y, z+1/2$; (ii) $-y, x-y-1, z$; (iii) $-x+y+1, -x+1, z$; (iv) $-x+1, -y, -z+1$; (v) $-x+1, -y, -z$; (vi) $x, y-1, z$; (vii) $-x+1, -y+1, z+1/2$; (viii) $x, y-1, -z+1/2$; (ix) $-x+1, -y+1, -z$; (x) $-x+y, -x, z$; (xi) $y, -x+y+1, -z$; (xii) $x, y+1, z$; (xiii) $-y, x-y, z$; (xiv) $x-y, x, -z$; (xv) $x, y, -z+1/2$; (xvi) $x, y, -z-1/2$; (xvii) $-y+1, x-y, z$; (xviii) $-x+y+1, -x, z$; (xix) $y, -x+y, -z$; (xx) $-x, -y, -z$; (xxi) $-x, -y, z-1/2$.