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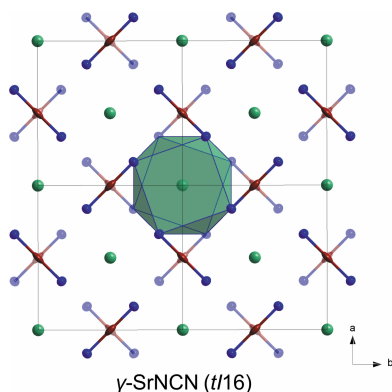
Synthesis and crystal structure of γ -SrNCN at 38 GPa

Lukas Brüning,^a Georg Krach,^b Pascal Lennert Jurzick,^a Wolfgang Schnick^b and Maxim Bykov^{a*}^aInstitute for Inorganic and Analytical Chemistry, Goethe University Frankfurt, Max-von-Laue-Strasse 7, 60438 Frankfurt am Main, Germany, and ^bDepartment of Chemistry, University of Munich (LMU), Butenandstrasse 5-13 (D), 81377 Munich, Germany. *Correspondence e-mail: maxim.bykov@chemie.uni-frankfurt.de

γ -Strontium carbodiimide, γ -SrNCN, was synthesized from a mixture of strontium subnitride (Sr_2N) and tetracyanoethylene (C_6N_4) at 38 (3) GPa in a laser-heated diamond anvil cell. Its crystal structure was solved and refined using synchrotron single-crystal X-ray diffraction. The new polymorph crystallizes in space group $I4/mcm$ (No. 140), where the Sr^{2+} and NCN^{2-} packing can be derived from the CsCl (B2) structure type. γ -SrNCN (*tI16*-SrNCN) is isostructural to *tI16*-BaNCN and represents the first high-pressure polymorph of SrNCN.

1. Chemical context

Inorganic carbodiimide salts are an interesting and well-established class of materials that can exhibit exciting optical, magnetic, and catalytic properties (Corkett *et al.*, 2024). Two polymorphs of SrNCN have been reported thus far. The first characterized polymorph of strontium carbodiimide, α -SrNCN (*oP16*-SrNCN, NaSCN structure type), was synthesized through a reaction of melamine ($\text{C}_3\text{N}_6\text{H}_6$) with strontium subnitride (Sr_2N) at 1123 K (Berger & Schnick, 1994). Polycrystalline β -SrNCN (*hR12*-SrNCN, β -NaN₃ structure type) was synthesized from SrCO_3 in liquid NH_3 (Wissmann, 2001), while crystals suitable for single-crystal X-ray diffraction were obtained by heating reactive fluxes of SrI_2 , NaCN and NaN₃ (2:1:1) at 1073 K, followed by slow cooling (Liao & Dronskowski, 2004). Krings *et al.* (2010) further expanded the range of synthetic routes to both α - and β -SrNCN and showed that β -SrNCN is the ground-state polymorph. α -SrNCN is used as a host lattice for Eu^{2+} doping, yielding an efficient orange-emitting phosphor (Krings *et al.*, 2011). Here, we report the synthesis of a high-pressure polymorph, γ -SrNCN (*tI16*-SrNCN), from a mixture of strontium subnitride (Sr_2N) and tetracyanoethylene (C_6N_4) at 38 (3) GPa. γ -SrNCN is isostructural to *tI16*-BaNCN, which is produced in a reaction of BaCO_3 in liquid NH_3 at ambient pressure and an elevated temperature of 1173 K (Masubuchi *et al.*, 2018). *tI16*-BaNCN remains stable upon compression up to 23 GPa. At higher pressures, it undergoes a symmetry-lowering phase transition to *mC16*-BaNCN, driven predominantly by tilting of the NCN^{2-} anions (Masubuchi *et al.*, 2022; Yamamoto *et al.*, 2026). There are a few other examples of high-pressure studies of carbodiimides (Solozhenko *et al.*, 2004; Glaser *et al.*, 2008; Möller *et al.*, 2018; Meinerzhagen *et al.*, 2024; Yang *et al.*, 2024). Furthermore, high-pressure and high-temperature conditions have proven the feasibility of synthesizing ternary nitridocarbonates with increased coordi-



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nation numbers of three and four for carbon (Brüning *et al.*, 2023; Aslandukov *et al.*, 2024).

2. Structural commentary

γ -SrNCN crystallizes in the space group $I4/mcm$ (No. 140, KN_3 structure type), where Sr, C, and N occupy the Wyckoff positions $4a$ (site symmetry 422), $4d$ (site symmetry $m.mm$) and $8h$ (site symmetry $m.2m$), respectively. The bond lengths and geometry of the NCN^{2-} anion are only weakly affected by pressure, with $d(C-N) = 1.222$ (13) Å in the reported structure compared to $d(C-N) = 1.232$ (5) Å in β -SrNCN at ambient pressure. In contrast to α -SrNCN and β -SrNCN, where Sr is sixfold coordinated in an octahedral environment with $d(Sr-N) = 2.600$ (8) – 2.657 (8) Å, γ -SrNCN features Sr in an eightfold tetragonal antiprismatic coordination with $d(Sr-N) = 2.460$ (4) Å. Similar to β -SrNCN, γ -SrNCN is built from stacked layers of Sr^{2+} cations and linear NCN^{2-} anions (Fig. 1). The main difference between the polymorphs arises from the rearrangement of NCN^{2-} units: in γ -SrNCN, they are oriented parallel to the Sr^{2+} layers rather than perpendicular to them like in β -SrNCN.

Within the NCN^{2-} layers, each linear unit is rotated by 90° in the ab plane with respect to the corresponding unit within the adjacent layer. Treating the NCN^{2-} unit as a single pseudo-atom would result in an octahedral coordination for both Sr and NCN^{2-} in β -SrNCN, while the coordination would be cubic for γ -SrNCN. Therefore, β -SrNCN can be derived from the NaCl (B1) structure type and γ -SrNCN packing follows the CsCl (B2) structure type, consistent with the pressure-coordination rule. A parallel can be drawn to the B1 to B2 phase transition in NaCl at 30 GPa (Bassett *et al.*, 1968)

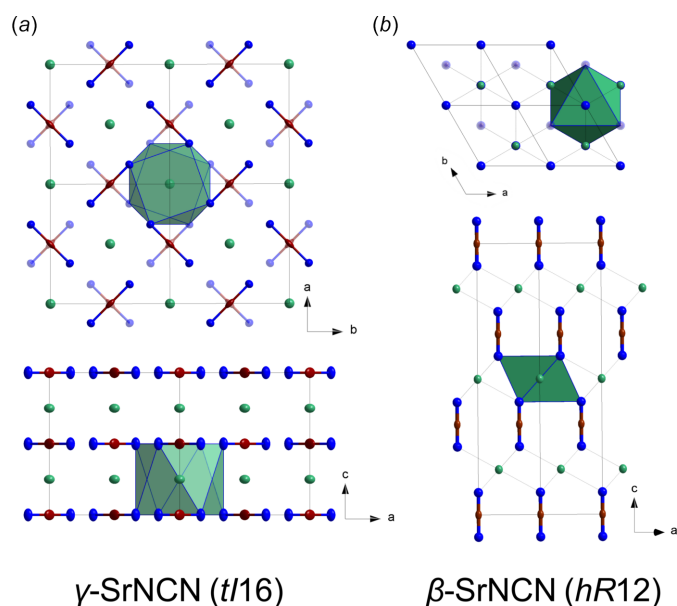


Figure 1

Crystal structure representations of (a) γ -SrNCN and (b) β -SrNCN along different crystallographic axes. Displacement ellipsoids are drawn at the 70% probability level. Sr, C and N atoms are colored green, brown and blue, respectively. Semitransparent atoms imply the next layer.

Table 1

Experimental details.

Crystal data	SrNCN
Chemical formula	127.65
M_r	Tetragonal, $I4/mcm$
Crystal system, space group	293
Temperature (K)	5.311 (2), 5.790 (6)
a, c (Å)	163.3 (2)
V (Å ³)	4
Z	Synchrotron, $\lambda = 0.2908$ Å
Radiation type	3.01
μ (mm ⁻¹)	0.001 × 0.001 × 0.001
Crystal size (mm)	
Data collection	Customized ω -circle diffractometer
Diffractometer	Multi-scan (<i>CrysAlis PRO</i> ; Rigaku OD, 2025)
Absorption correction	0.150, 1.000
T_{min}, T_{max}	366, 109, 70
No. of measured, independent and observed [$I > 2\sigma(I)$] reflections	0.072
R_{int}	1.006
($\sin \theta/\lambda$) _{max} (Å ⁻¹)	
Refinement	
$R[F^2 > 2\sigma(F^2)], wR(F^2), S$	0.060, 0.156, 1.15
No. of reflections	109
No. of parameters	10
$\Delta\rho_{max}, \Delta\rho_{min}$ (e Å ⁻³)	1.97, -1.69

Computer programs: *CrysAlis PRO* (Rigaku OD, 2025), *SHELXT2014/5* (Sheldrick, 2015a), *SHELXL2014/7* (Sheldrick, 2015b) and *OLEX2* (Dolomanov *et al.*, 2009).

and to the polymorphism of NaN_3 , for which the isostructural high-pressure polymorph $tI16$ - NaN_3 was described (Pulham *et al.*, 2014). The results also align well with the pressure homologue rule, since CsCl (B2) packing can be achieved in $tI16$ -BaNCN at ambient pressure (Masubuchi *et al.*, 2018).

3. Synthesis and crystallization

Strontium subnitride (Sr_2N) was synthesized via direct reaction of strontium metal with N_2 gas at 1273 K, as described in the literature (Reckeweg & DiSalvo, 2002). A 20 μm piece of Sr_2N was embedded in tetracyanoethylene (C_6N_4 , Thermo Fischer, purity > 98%), compressed to 38 (3) GPa and laser-heated in a diamond anvil cell (BX90 body design; Boehler-Almax type diamonds with a conical aperture of 70° and 200 μm culet size) with a Nd:YAG laser ($\lambda = 1064$ nm, $T > 1500$ K, 4 s heating time). The same synthesis approach and detailed description of data evaluation can be found in previous works on ternary nitridocarbonates (Brüning *et al.*, 2023, 2025; Ranieri *et al.*, 2025; Jurzick *et al.*, 2026). Pressure was determined using the pressure–frequency relationship of the stressed diamond Raman band (Akahama & Kawamura, 2006). The reaction product, γ -SrNCN, was polycrystalline with submicron grain sizes, as indicated by XRD mapping of the sample chamber.

4. Refinement

Crystal data, data collection and structure refinement details are summarized in Table 1. The reaction product was studied by means of synchrotron single-crystal X-ray diffraction at the

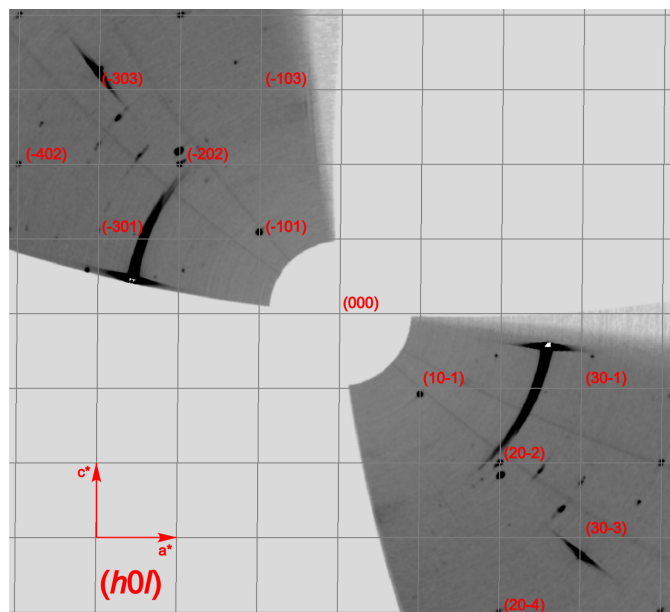


Figure 2
($h0l$)-reciprocal lattice plane of γ -SrNCN (No. 140 $I4/mcm$) at 38 (3) GPa with indexed reflections fulfilling (hkl): $h + k + l = 2n$ for I -centering and ($0kl$): $k, l = 2n$ [\equiv ($h0l$): $h, l = 2n$] for a c -glide plane perpendicular to $[100]$ ($\equiv [010]$).

extreme conditions beamline P02.2 at Deutsches Elektronen Synchrotron (PetraIII, DESY). The X-ray beam had a full width at half maximum of $\sim 2 \mu\text{m}$ and a wavelength of 0.2908 \AA , and the X-ray diffraction data were measured using a PerkinElmer XRD1621 2D flat panel detector. To obtain single-crystal datasets, the diamond anvil cell was rotated around the vertical ω axis within a range of $\pm 32^\circ$. Diffraction data were acquired in $0.5^\circ \omega$ steps with an exposure time of 4s per $^\circ$. Data reduction was performed using *CrysAlis^{PRO}* software package. Since the dataset contains diffraction data from multiple crystallites with different orientations, the algorithm DaFi (Aslandukov *et al.*, 2022) was used to group and extract the orientation matrices from individual crystalline domains. The most prominent domain was used for integration and the resulting hkl file was then used for structure solution with *SHELXT* (Sheldrick 2015a) and refinement with *SHELXL* (Sheldrick 2015b) within the *OLEX2-1.5* interface (Dolomanov *et al.*, 2009). The limited opening of the diamond anvil cell results in reduced completeness of the datasets. As a standard procedure, we carefully examine reconstructed precession images of each dataset to check for missing superlattice reflections and to cross-check the choice of the space-group symmetry. In the case of γ -SrNCN, the systematic absences are consistent with the $I4/mcm$ space group. It can be shown that ($0kl$): $k, l = 2n$ [\equiv ($h0l$): $h, l = 2n$] for a c -glide plane perpendicular to $[100]$ ($\equiv [010]$) in a tetragonal body-centered Bravais lattice holds (Fig. 2).

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Synthesis and crystal structure of γ -SrNCN at 38 GPa

Lukas Brüning, Georg Krach, Pascal Lennert Jurzick, Wolfgang Schnick and Maxim Bykov

Computing details

 γ -Strontium carbodiimide

Crystal data

SrNCN	$D_x = 5.191 \text{ Mg m}^{-3}$
$M_r = 127.65$	Synchrotron radiation, $\lambda = 0.2908 \text{ \AA}$
Tetragonal, $I4/mcm$	Cell parameters from 94 reflections
$a = 5.311 (2) \text{ \AA}$	$\theta = 2.2\text{--}14.7^\circ$
$c = 5.790 (6) \text{ \AA}$	$\mu = 3.01 \text{ mm}^{-1}$
$V = 163.3 (2) \text{ \AA}^3$	$T = 293 \text{ K}$
$Z = 4$	Irregular, dull dark gray
$F(000) = 232$	$0.001 \times 0.001 \times 0.001 \text{ mm}$

Data collection

Customized ω -circle diffractometer	$T_{\min} = 0.150$, $T_{\max} = 1.000$
Radiation source: synchrotron, PETRAIII, Beamline P02.2	366 measured reflections
Synchrotron monochromator	109 independent reflections
Detector resolution: $5.0 \text{ pixels mm}^{-1}$	70 reflections with $I > 2\sigma(I)$
ω scans	$R_{\text{int}} = 0.072$
Absorption correction: multi-scan (CrysAlisPro; Rigaku OD, 2025)	$\theta_{\max} = 17.0^\circ$, $\theta_{\min} = 3.6^\circ$
	$h = -7 \rightarrow 7$
	$k = -8 \rightarrow 7$
	$l = -7 \rightarrow 8$

Refinement

Refinement on F^2	0 restraints
Least-squares matrix: full	Primary atom site location: dual
$R[F^2 > 2\sigma(F^2)] = 0.060$	$w = 1/[\sigma^2(F_o^2) + (0.0651P)^2]$
$wR(F^2) = 0.156$	where $P = (F_o^2 + 2F_c^2)/3$
$S = 1.15$	$(\Delta/\sigma)_{\max} < 0.001$
109 reflections	$\Delta\rho_{\max} = 1.97 \text{ e \AA}^{-3}$
10 parameters	$\Delta\rho_{\min} = -1.69 \text{ e \AA}^{-3}$

Special details

Experimental. X-ray diffraction was measured at 38 (3) GPa on synthesis products, produced by laser-heating in a diamond anvil cell with an opening angle of 70° . Tetracyanoethylene was used as the pressure-transmitting medium. Pressure was determined using the pressure-frequency relationship of the stressed diamond Raman band.

Geometry. All esds (except the esd in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell esds are taken into account individually in the estimation of esds in distances, angles and torsion angles; correlations between esds in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell esds is used for estimating esds involving l.s. planes.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters (\AA^2)

	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{iso}}^*/U_{\text{eq}}$
Sr01	0.5000	0.5000	0.2500	0.0118 (5)
C1	0.5000	0.0000	0.0000	0.012 (3)
N1	0.3373 (18)	0.1627 (18)	0.0000	0.013 (2)

Atomic displacement parameters (\AA^2)

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Sr01	0.0132 (6)	0.0132 (6)	0.0092 (11)	0.000	0.000	0.000
C1	0.012 (5)	0.012 (5)	0.013 (11)	-0.008 (6)	0.000	0.000
N1	0.009 (3)	0.009 (3)	0.019 (8)	0.001 (4)	0.000	0.000

Geometric parameters (\AA , $^\circ$)

Sr01—Sr01 ⁱ	2.895 (3)	C1—Sr01 ^{ix}	3.0245 (11)
Sr01—Sr01 ⁱⁱ	2.895 (3)	C1—Sr01 ^x	3.0245 (11)
Sr01—C1 ⁱⁱⁱ	3.0245 (11)	C1—Sr01 ^{xi}	3.0245 (11)
Sr01—C1	3.0245 (11)	C1—Sr01 ^{vii}	3.0245 (11)
Sr01—N1 ^{iv}	2.460 (4)	C1—Sr01 ^{xii}	3.0245 (11)
Sr01—N1 ^v	2.460 (4)	C1—Sr01 ⁱ	3.0245 (11)
Sr01—N1	2.460 (4)	C1—Sr01 ^{xiii}	3.0245 (11)
Sr01—N1 ^{vi}	2.460 (4)	C1—N1	1.222 (13)
Sr01—N1 ⁱ	2.460 (4)	C1—N1 ^{xii}	1.222 (13)
Sr01—N1 ^{vii}	2.460 (4)	N1—Sr01 ^{xi}	2.460 (4)
Sr01—N1 ⁱⁱⁱ	2.460 (4)	N1—Sr01 ^{vii}	2.460 (4)
Sr01—N1 ^{viii}	2.460 (4)	N1—Sr01 ⁱ	2.460 (4)
Sr01 ⁱⁱ —Sr01—Sr01 ⁱ	180.0	N1 ^{viii} —Sr01—N1 ^{vi}	86.5 (5)
Sr01 ⁱⁱ —Sr01—C1	118.59 (3)	N1 ^{iv} —Sr01—N1 ^v	149.1 (6)
Sr01 ⁱ —Sr01—C1 ⁱⁱⁱ	118.59 (3)	N1 ⁱⁱⁱ —Sr01—N1 ^{vii}	69.74 (7)
Sr01 ⁱ —Sr01—C1	61.41 (3)	N1—Sr01—N1 ^{vi}	69.74 (7)
Sr01 ⁱⁱ —Sr01—C1 ⁱⁱⁱ	61.41 (3)	N1 ^{iv} —Sr01—N1 ⁱⁱⁱ	80.5 (2)
C1—Sr01—C1 ⁱⁱⁱ	180.0	Sr01—C1—Sr01 ^{xi}	103.24 (2)
N1 ^{viii} —Sr01—Sr01 ⁱ	126.04 (7)	Sr01 ^{xii} —C1—Sr01 ^{vii}	103.24 (2)
N1 ^{iv} —Sr01—Sr01 ⁱⁱ	126.04 (7)	Sr01 ^{vii} —C1—Sr01 ^x	180.0
N1 ⁱⁱⁱ —Sr01—Sr01 ⁱⁱ	53.96 (7)	Sr01 ^{xi} —C1—Sr01 ^{vii}	57.19 (5)
N1—Sr01—Sr01 ⁱ	53.96 (7)	Sr01 ^{ix} —C1—Sr01 ^x	57.19 (5)
N1—Sr01—Sr01 ⁱⁱ	126.04 (7)	Sr01 ^{xiii} —C1—Sr01 ^x	103.24 (2)
N1 ^{iv} —Sr01—Sr01 ⁱ	53.96 (7)	Sr01 ^{ix} —C1—Sr01 ^{vii}	122.81 (5)
N1 ^{vi} —Sr01—Sr01 ⁱ	53.96 (7)	Sr01 ^{ix} —C1—Sr01 ^{xi}	180.0
N1 ⁱ —Sr01—Sr01 ⁱ	53.96 (7)	Sr01 ⁱ —C1—Sr01 ^{vii}	103.24 (2)
N1 ^{viii} —Sr01—Sr01 ⁱⁱ	53.96 (7)	Sr01—C1—Sr01 ^x	103.24 (2)
N1 ⁱ —Sr01—Sr01 ⁱⁱ	126.04 (7)	Sr01 ^{xii} —C1—Sr01 ^x	76.76 (2)
N1 ^{vii} —Sr01—Sr01 ⁱⁱ	53.96 (7)	Sr01—C1—Sr01 ^{xii}	180.0
N1 ^{vi} —Sr01—Sr01 ⁱⁱ	126.04 (7)	Sr01—C1—Sr01 ^{ix}	76.76 (2)
N1 ^{vii} —Sr01—Sr01 ⁱ	126.04 (7)	Sr01—C1—Sr01 ^{vii}	76.76 (2)

N1 ^v —Sr01—Sr01 ⁱ	126.04 (7)	Sr01 ^{xi} —C1—Sr01 ^{xiii}	103.24 (2)
N1 ^v —Sr01—Sr01 ⁱⁱ	53.96 (7)	Sr01 ⁱ —C1—Sr01 ^x	76.76 (2)
N1 ⁱⁱⁱ —Sr01—Sr01 ⁱ	126.04 (7)	Sr01 ^{xii} —C1—Sr01 ^{xiii}	57.19 (5)
N1 ⁱ —Sr01—C1	110.97 (18)	Sr01 ^{xiii} —C1—Sr01 ^{vii}	76.76 (2)
N1 ^{vii} —Sr01—C1 ⁱⁱⁱ	91.5 (2)	Sr01—C1—Sr01 ⁱ	57.19 (5)
N1 ⁱⁱⁱ —Sr01—C1	157.1 (3)	Sr01 ^{xi} —C1—Sr01 ^{xii}	76.76 (2)
N1 ⁱⁱⁱ —Sr01—C1 ⁱⁱⁱ	22.9 (3)	Sr01 ^{ix} —C1—Sr01 ⁱ	103.24 (2)
N1 ^{viii} —Sr01—C1 ⁱⁱⁱ	53.8 (3)	Sr01 ^{xi} —C1—Sr01 ^x	122.81 (5)
N1 ^{vii} —Sr01—C1	88.5 (2)	Sr01 ^{xi} —C1—Sr01 ⁱ	76.76 (2)
N1 ⁱ —Sr01—C1 ⁱⁱⁱ	69.03 (18)	Sr01—C1—Sr01 ^{xiii}	122.81 (5)
N1 ^{vi} —Sr01—C1	53.8 (3)	Sr01 ^{xii} —C1—Sr01 ⁱ	122.81 (5)
N1 ^{viii} —Sr01—C1	126.2 (3)	Sr01 ^{xiii} —C1—Sr01 ⁱ	180.0
N1 ^{vi} —Sr01—C1 ⁱⁱⁱ	126.2 (3)	Sr01 ^{ix} —C1—Sr01 ^{xiii}	76.76 (2)
N1—Sr01—C1 ⁱⁱⁱ	157.1 (3)	Sr01 ^{ix} —C1—Sr01 ^{xii}	103.24 (2)
N1 ^v —Sr01—C1	69.03 (18)	N1 ^{xii} —C1—Sr01 ^{vii}	128.380 (11)
N1—Sr01—C1	22.9 (3)	N1 ^{xii} —C1—Sr01 ⁱ	128.380 (11)
N1 ^{iv} —Sr01—C1 ⁱⁱⁱ	88.5 (2)	N1—C1—Sr01 ^{xii}	128.380 (11)
N1 ^{iv} —Sr01—C1	91.5 (2)	N1—C1—Sr01 ^x	128.380 (11)
N1 ^v —Sr01—C1 ⁱⁱⁱ	110.97 (18)	N1 ^{xii} —C1—Sr01 ^{ix}	51.620 (11)
N1 ^{viii} —Sr01—N1 ⁱ	80.5 (2)	N1 ^{xii} —C1—Sr01 ^{xii}	51.620 (11)
N1 ^{vi} —Sr01—N1 ^v	80.5 (2)	N1 ^{xii} —C1—Sr01	128.380 (11)
N1—Sr01—N1 ^v	86.5 (5)	N1—C1—Sr01 ^{vii}	51.620 (11)
N1 ⁱ —Sr01—N1 ^v	138.9 (5)	N1—C1—Sr01 ^{xiii}	128.380 (11)
N1 ^{viii} —Sr01—N1 ⁱⁱⁱ	69.74 (7)	N1—C1—Sr01 ⁱ	51.620 (11)
N1 ^{viii} —Sr01—N1 ^{vii}	107.91 (13)	N1 ^{xii} —C1—Sr01 ^{xi}	128.380 (11)
N1 ^{vii} —Sr01—N1 ^v	69.74 (7)	N1—C1—Sr01 ^{ix}	128.380 (11)
N1—Sr01—N1 ⁱ	107.91 (13)	N1 ^{xii} —C1—Sr01 ^x	51.620 (11)
N1 ^{viii} —Sr01—N1 ^{iv}	138.9 (5)	N1—C1—Sr01	51.620 (11)
N1—Sr01—N1 ^{iv}	69.74 (7)	N1 ^{xii} —C1—Sr01 ^{xiii}	51.620 (11)
N1 ^{viii} —Sr01—N1 ^v	69.74 (7)	N1—C1—Sr01 ^{xi}	51.620 (11)
N1 ^{iv} —Sr01—N1 ⁱ	69.74 (7)	N1 ^{xii} —C1—N1	180.0 (13)
N1 ^{iv} —Sr01—N1 ^{vii}	86.5 (5)	Sr01—N1—Sr01 ⁱ	72.09 (13)
N1 ⁱⁱⁱ —Sr01—N1 ^v	107.91 (13)	Sr01 ^{xi} —N1—Sr01	149.1 (6)
N1 ^{iv} —Sr01—N1 ^{vi}	107.91 (13)	Sr01 ^{xi} —N1—Sr01 ⁱ	99.5 (2)
N1 ⁱⁱⁱ —Sr01—N1 ⁱ	86.5 (5)	Sr01 ^{xi} —N1—Sr01 ^{vii}	72.09 (13)
N1 ⁱⁱⁱ —Sr01—N1 ^{vi}	149.1 (6)	Sr01—N1—Sr01 ^{vii}	99.5 (2)
N1 ^{viii} —Sr01—N1	149.1 (6)	Sr01 ^{vii} —N1—Sr01 ⁱ	149.1 (6)
N1 ^{vii} —Sr01—N1 ^{vi}	138.9 (5)	C1—N1—Sr01 ⁱ	105.5 (3)
N1 ^{vii} —Sr01—N1 ⁱ	149.1 (6)	C1—N1—Sr01 ^{xi}	105.5 (3)
N1 ⁱ —Sr01—N1 ^{vi}	69.74 (7)	C1—N1—Sr01 ^{vii}	105.5 (3)
N1—Sr01—N1 ⁱⁱⁱ	138.9 (5)	C1—N1—Sr01	105.5 (3)
N1—Sr01—N1 ^{vii}	80.5 (2)		
Sr01—C1—N1—Sr01 ^{xi}	180.000 (1)	Sr01 ^{ix} —C1—N1—Sr01	0.000 (1)
Sr01 ^{xii} —C1—N1—Sr01 ^{vii}	75.25 (6)	Sr01 ^{xiii} —C1—N1—Sr01 ^{xi}	-75.25 (6)
Sr01 ^{xiii} —C1—N1—Sr01 ⁱ	180.0	Sr01 ^{xi} —C1—N1—Sr01	180.0
Sr01 ^{xi} —C1—N1—Sr01 ^{vii}	75.25 (6)	Sr01 ^{xiii} —C1—N1—Sr01 ^{vii}	0.0
Sr01—C1—N1—Sr01 ⁱ	75.25 (6)	Sr01 ^{xii} —C1—N1—Sr01	180.0

Sr01 ^{xii} —C1—N1—Sr01 ⁱ	-104.75 (6)	Sr01 ⁱ —C1—N1—Sr01 ^{xi}	104.75 (6)
Sr01 ^{ix} —C1—N1—Sr01 ^{vii}	-104.75 (6)	Sr01 ^{xiii} —C1—N1—Sr01	104.75 (6)
Sr01 ^{ix} —C1—N1—Sr01 ^{xi}	180.0	Sr01 ^{ix} —C1—N1—Sr01 ⁱ	75.25 (6)
Sr01 ⁱ —C1—N1—Sr01 ^{vii}	180.0	Sr01 ⁱ —C1—N1—Sr01	-75.25 (6)
Sr01 ^x —C1—N1—Sr01 ^{vii}	180.0	Sr01 ^{vii} —C1—N1—Sr01 ^{xi}	-75.25 (6)
Sr01 ^{xi} —C1—N1—Sr01 ⁱ	-104.75 (6)	Sr01 ^{vii} —C1—N1—Sr01	104.75 (6)
Sr01 ^{xii} —C1—N1—Sr01 ^{xi}	0.000 (1)	Sr01 ^{vii} —C1—N1—Sr01 ⁱ	180.0
Sr01 ^x —C1—N1—Sr01 ⁱ	0.000 (1)	Sr01 ^x —C1—N1—Sr01	-75.25 (6)
Sr01—C1—N1—Sr01 ^{vii}	-104.75 (6)	Sr01 ^x —C1—N1—Sr01 ^{xi}	104.75 (6)

Symmetry codes: (i) $-x+1, -y+1, -z$; (ii) $-x+1, -y+1, -z+1$; (iii) $-y+1/2, x+1/2, z+1/2$; (iv) $y, -x+1, -z$; (v) $y+1/2, -x+1/2, -z+1/2$; (vi) $-y+1, x, z$; (vii) $-x+1/2, -y+1/2, -z+1/2$; (viii) $x+1/2, y+1/2, z+1/2$; (ix) $-x+3/2, -y+1/2, -z+1/2$; (x) $x+1/2, y-1/2, z-1/2$; (xi) $x-1/2, y-1/2, z-1/2$; (xii) $-x+1, -y, -z$; (xiii) $x, y-1, z$.