Electronic structure of Schiff-base peroxo{2,2'-[1,2phenylenebis(nitrilomethanylylidene)]bis(6-methoxyphenolato) { titanium(IV) monohydrate: a possible model structure of the reaction center for the theoretical study of hemoglobin

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An extensive characterization of [Ti(C₂₂H₁₈N₂O₆)]·H₂O was performed by topological analysis according to Bader's quantum theory of atoms in molecules (QTAIM) from the experimentally (multipole model) and theoretically (DFT) determined electron density. To the best of our knowledge, this study is the first example of an experimental electronic structure of a coordination compound in which a peroxo anion is bonded to a 3d central atom. The titanium coordination polyhedron could be described as a deformed tetrahedral pyramid if the midpoint of the peroxide O-O bond (side-on mode) is considered to be in the quasi-apical position. According to the multipole model (MM) results, the titanium atom has a positive QTAIM charge of 2.05 e⁻ which does not correspond to the formal Ti (IV) oxidation state. On the other hand, the peroxo oxygen atoms O(1) and O(2) have MM QTAIM charges of -0.27 and -0.12, respectively. This asymmetric charge density distribution on the peroxo oxygens is in agreement with the distorted orientation of the O₂ moiety with respect to the titanium atom. Despite the fact that the overall MM charge of the O₂ moiety is more remote from the formal -2 charge than from neutral O_2 , the O-Odistance remains close to that in the peroxo O_2^{2-} anion. In the case of DFT results, the titanium atom charge is also found to be close to +2, the O_2^{x-} moiety charge is around -1, the optimized O-O distance is shorter by only *ca* 0.04 Å than the experimental value of 1.5005 (16) Å, and the DFT *d*-populations on titanium are found to be lower than the experimental MM value. This study is the first experimental electronic structure of a transition metal peroxo complex.

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1. Introduction

All living organisms requiring molecular oxygen for life mediate four-electron reduction of oxygen to water (Valko et al., 2004). In the course of the reduction process, the energy formed is utilized by aerobic organisms maintaining life on the Earth. Molecular oxygen is in triplet ground state with two parallel unpaired electrons (S = 1) which represents the most stable oxygen form. The first step of the reduction cascade, representing reduction of molecular oxygen to superoxide radical anion is a rather unfavorable, endergonic reaction $(\sim 33000 \text{ J mol}^{-1})$ (Valko *et al.*, 2005). The molecular oxygen biradical has two parallel electrons in antibonding π orbitals and therefore its reactions with organic molecules, in which all electrons are paired and in closed-shell systems, are spin forbidden. To overcome thermodynamic and electronic restrictions in the process of reduction of molecular oxygen,





nature has evolved a variety of metallo-enzymes to store and transport molecular oxygen as well as catalyze its conversion to more reduced forms.

In their structure metallo-enzymes contain integrally anchored transition metal ions with unpaired *d* electrons (Rebilly *et al.*, 2015). Paramagnetic metal ions in various electronic states can activate molecular oxygen in its ground state and make the process of reduction more feasible. Interaction of molecular oxygen with metal centers of metalloproteins or coordination compounds containing transition metals changes the thermodynamics as well as the kinetics of oxygen reduction. Upon interaction of metal ions with molecular oxygen a variety of intermediates such as superoxo, hydroperoxo and oxo-species are formed (Valko *et al.*, 1995). These metal–oxygen adducts are usually less reactive and more stable than typical organic radicals such as reactive oxygen species (ROS), or reactive nitrogen species (RNS).

In the past two to three decades, titanium has been increasingly used in materials that improve the life quality of humans. Titanium is a key component of prosthetics and therefore this element is in direct contact with biological fluids and/or tissues. Thus, the interaction of titanium with physiological target molecules of various molecular weights and biological functions may occur extensively. This raises the question of what types of interaction may develop between the metal ion (in various oxidation states) and surrounding biological tissues. Such interactions may play an important role in the activation of signal-transduction pathways, activation of enzymes and expression of genes which in turn may explain the clinical symptoms described in clinical investigations (Dakanali *et al.*, 2003).

Interaction of titanium complexes with dioxygen is of importance for better understanding the nature of metaloxygen interactions, the reversible reaction of carbon dioxide with metal oxides, development of more effective titanium dioxygen species with improved photocatalytic activity and other properties (De Lile *et al.*, 2017).



The interaction of dioxygen with titanium is a result of the interaction between the oxygen antibonding π^* orbital and the non-bonding *d* orbitals of the metal ion. Generally, there are four possible modes of dioxygen binding to a metal center: *end-on*, *side-on* and *bridge* (Scheme 1). Titanium capable of binding peroxides can activate them toward oxidation of a variety of substrates and consequently decompose them

(DiPasquale *et al.*, 2002). Interest related to Ti(IV)-peroxo species has been associated with reactive superoxide radical anion traps and various reactive oxygen species formed during inflammatory processes (Tengvall *et al.*, 1991).

In the Cambridge Structural database the crystal structures of transition metals (M) with one, two or three peroxo groups (*side-on* mode, different geometries) may be found as follows:

(i) one peroxo group for M = V (80 structures), Mo (40 structures), Ti (33 structures), Ru (25 structures), Rh, W, Ir (19 structures), Nb (6 structures), and Os, Mn, Co, Ni, Ta, Hf and Re (one structure);

(ii) two peroxo groups for M = V, Nb, Mo, W, Re, Hf and Ir; (iii) three peroxo groups for M = V, Nb and Ta.

From 33 Ti crystal structures with one coordinated peroxo group, we will briefly point out those that are *side-on* coordinated for comparison purposes with the title compound. The crystal structure of $[Sr(H_2O)_7][Ti(O_2)(edta)] \cdot H_2O$ (KOGHUK, Deng & Zhou, 2008) has the peroxo-group distances Ti-O(1), Ti-O(2) and O(1)-O(2) of 1.833 (2), 1.844 (2) and 1.453 (3) Å, respectively. In crystal structures of titanium ethylenediamine with chromophore $[Ti(O_2)O_3N_2]$ [VINGAB, VINGEF, VINGIJ and VINGUV (Zhou *et al.*, 2007)] the analogous Ti-O(ethylenediamine) distances are in the interval 1.948–2.086 Å and those of Ti(1)–N(ethylenediamine) are in the interval 2.192–2.300 Å.

Through the experimental electron density, chemical bonding can be understood from experimental and theoretical points of view. More than 99.9% of single-crystal structure determinations are based on the spherical atom model. These studies are able to determine the bond distances and angles between atoms in the molecule, as well as all interatomic interactions [of course with small systematic errors due to the independent atom model (IAM)]. On the other hand, using the Hansen-Coppens multipole formalism (Hansen & Coppens, 1978) for accurate diffraction data with satisfactory resolution, providing all necessary corrections, it is possible to obtain a valence electron distribution that reflects bonding properties and interactions in the studied molecule. Despite the low suitability factor (Coppens, 1997), recently published papers of experimental electronic structures of 3d-coordination compounds show reasonable results (Schmøkel et al., 2013; Herich et al., 2018a,b; Fukin et al., 2019; Gao et al., 2019; Scatena et al., 2019).

Targeted preparation of compounds with desirable properties requires a deep understanding of the relationship between the structure and properties of the compounds studied, which are closely related to their chemical bond. The goal of our study is to elucidate weakening of the O–O bond in the studied peroxo complex which may facilitate a subsequent redox reaction of the coordinated peroxo group.

2. Experimental

2.1. Material and methods

2.1.1. Synthesis and crystal growth. All chemicals were purchased commercially and were used as received without

further purification. The synthesis was carried out in three steps. In the first step, we prepared the Schiff base from ovanillin (0.304 g, 2 mmol, o-vanillin 99%, Alfa Aesar) and ophenylendiamine (0.108 g, 1 mmol, o-phenylenediamine 99%, Sigma-Aldrich) in 50 ml methanol [methanol (p.a.) was a product of CENTRALCHEM]. The Schiff base solution was stirred for 30 min. In the second step, titanium (IV) butoxide [0.340 g, 1 mmol (in a 5% surplus), titanium (IV) butoxide 97%, reagent grade, Sigma-Aldrich] was added to the Schiff base solution with vigorous stirring. The vellow Schiff base solution turned dark orange. The precipitated TiO₂ was filtered from the solution. Finally, we added H_2O_2 [0.034 g, 1 mmol (0.11 g, 30% solution), hydrogen peroxide 30%, reagent grade ISO, Sigma-Aldrich] and the dark orange solution turned light orange. The orange crystals dropped out of solution after 1 d in the refrigerator. After crystallization a single crystal suitable for X-ray was selected.

2.1.2. Data collection. A high-quality yellow rod-shaped single crystal with the dimensions $0.150 \times 0.050 \times 0.045$ mm was measured on a Eulerian four-circle diffractometer Stoe STADIVARI with a Dectris Pilatus 300 K detector, Incoatec IµS Ag microfocus source (Ag $K\alpha$, $\lambda = 0.56083$ Å) at 100 K using a nitrogen gas open-flow Cobra cooling system from Oxford Cryosystems. Two detector positions for 64 omega scans ($2\theta = 4.5$ and 89.3°) with a 0.5° frame width were used. The exposure time was 200 s. The maximum resolution reached at this experimental setting was d = 0.399 Å and $\sin(\theta)/\lambda = 1.253 \text{ Å}^{-1}$. The data reduction was performed using X-Area Integrate (version 1.73.1) and X-Area X-Red32 (version 1.65.0.0; Stoe & Cie, 2018). For absorption correction a crystal-shape model with eight faces was employed. The average redundancy was 9.9, R_{int} and R_{σ} were 0.0696 and 0.0307, respectively. From the data reduction we obtained direction cosines and TBAR (distance of the primary and diffracted beam through the crystal) first as described previously (Kožíšek et al., 2002; Herich et al., 2018a). Details of the X-ray diffraction experiment conditions and the crystallographic data are given in Table 1. As the symmetryequivalent data were collected with a different value of TBAR, all non-averaged data were used in the refinements.

2.2. Electron density refinements

The structure was solved by the dual-space algorithm implemented in *SHELXT* (Sheldrick, 2015*a*). The IAM was refined using *SHELXL* (Sheldrick, 2015*b*) and the graphical user interface *Olex2* (Dolomanov *et al.*, 2009). For *MM* refinement the Hansen–Coppens model (Hansen & Coppens, 1978) was used. The total atomic density $\rho(\mathbf{r})$ in this approach is divided into three contributions:

$$\rho(\mathbf{r}) = P_{\text{core}} \rho_{\text{core}}(r) + P_{\text{val}} \kappa^3 \rho_{\text{val}}(\kappa r) + \sum_{l=0}^{l_{\text{max}}} \kappa'^3 R_l(\kappa' r) \sum_{m=0}^{l} P_{lm\pm} Y_{lm\pm}(\Omega).$$
(1)

The first two components describe the spherical *core* and spherical *valence* electron density (Hansen & Coppens, 1978) and the third term describes the aspherical deformation of the

Table 1Experimental details.

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Crystal data	
Chemical formula	$[Ti(C_{22}H_{18}N_2O_6)]\cdot H_2O$
Formula weight, Z	472.30, 2
F(000)	488
Temperature (K)	100.0 (1)
Crystal size (mm)	$0.145 \times 0.192 \times 0.476$
a (Å)	7.4798 (2)
b (Å)	12.2525 (4)
$c(\dot{A})$	12.6131 (4)
α (°)	115.637 (2)
β (°)	91.030 (3)
γ (°)	103.625 (3)
$V(Å^3)$	1003.29 (6)
Space group	No. 2, <i>P</i> 1
Wavelength (Å)	0.56083
$\mu (\mathrm{mm}^{-1})$	0.254
Scan type	Ω scans
Crystal size (mm)	$0.045 \times 0.082 \times 0.150$
Max $\sin(\theta/\lambda)$ (Å ⁻¹)	1.253
Range of indices for h	-18, +17=
Range of indices for k	± 30
Range of indices for <i>l</i>	± 31
No. of measured diffractions	80862
Redundancy (all)	10.08
$R_{\rm int}$ (for resolution 0.65 Å)	0.0696
$R(\sigma)$ (for resolution 0.65 Å)	0.0307
Crystal to detector (mm)	40
X-ray tube (kV), μA	50, 880
Multipole refinement on F^2	
R(F), $wR(F)$, GOOF	0.0385, 0.0172, 0.9098
$R(F^2), wR(F^2), GOOFw$	0.0486, 0.0329, 0.9122
N _{ref} /N _v	51.6539
$\Delta \rho_{\rm max}, \Delta \rho_{\rm min} ({ m e}{ m \AA}^{-3})$	2.38, -0.82

valence electron density. R_l are normalized Slater-type radial functions and $Y_{lm\pm}$ are the density normalized real spherical harmonics. The parameters κ and κ' are responsible for contraction/expansion of the spherical and aspherical valence parts.

MM refinement calculations were based on F^2 refinements using the XD2016 (Volkov et al., 2016) suite of programs and the low-temperature (100 K) X-ray diffraction data. The leastsquares procedure accounted only for reflections with I > $3\sigma(I)$ using the Su-Coppens (Su & Coppens, 1998) (SCM) wavefunctions databank. Details on the refinements are provided in S1 of the supporting information. An error analysis revealed that there is quite a large fluctuation of the scale factors *versus* $\sin(\theta)/\lambda$. The residual density calculated by fast Fourier synthesis (XDFFT) for all diffractions is 3.43 e Å⁻³ at 0.03 Å from the titanium atom and -1.21 e Å⁻³ at 0.48 Å from the titanium atom with a mean value of 0.155 e $Å^{-3}$. We introduced 19 scale factors into the multipole refinement, one for each group, as suggested by Niepötter et al. (2015). After the complete procedure of multipole refinement the residual density decreased to 2.38 e Å⁻³ at 0.02 Å from the titanium atom and -0.82 e Å⁻³ at 0.45 Å from the titanium atom with a mean value of 0.134 e $Å^{-3}$. As integration in the atomic basin gives much lower charges for peroxooxygen atoms, different scattering curves (neutral for O_2^{2-} and O⁻ for other oxygen atoms) were introduced. Multipole refinement was repeated with slightly lower R values obtained

(3.8622% versus 3.8544%). Small changes were observed for charges of peroxo-oxygen atoms, they are more negative and other oxygen atoms less negative (see Table S4). Also atomic volumes for non-peroxo oxygen atoms are larger. At the end of the multipolar refinement we found that it is not necessary to refine the secondary extinction.

The fractal plot of the residual density (Meindl & Henn, 2008) has a symmetrical shape for the entire $\sin\theta/\lambda$ range of the data set with $\rho_{\rm min} = -0.77$ e Å⁻³ and $\rho_{\rm max} = 0.207$ e Å⁻³ (see Fig. S1 of the supporting information).

The normal probability distribution plot (Abrahams & Keve, 1971; Farrugia, 2012) shows a fairly good agreement with the assumed shape (Fig. S2 of the supporting information). The slope is 45° , the function goes through the origin and is linear in the interval from -3 to 3 (Abrahams & Keve, 1971). The variation of the scale factor with respect to the resolution is about 8% higher for the last group (see Fig. S3 (Farrugia, 2012) in the supporting information). It could be said that the error analysis has affirmed a good agreement between the experimental and calculated structure factors.

2.3. Quantum chemical calculations

Geometry optimization of the neutral complex under study in singlet ground state starting from the X-ray structure and of the O_2^q molecule, with charges q = 0, -1 and -2 in various spin states, was performed by employing the B3LYP hybrid functional (Becke, 1988; Lee et al., 1988; Vosko et al., 1980; Stephens et al., 1994) with Grimme's D3 dispersion corrections (Grimme et al., 2010) and 6-311+G* basis sets from the Gaussian library (Frisch et al., 2016) for all atoms. Alternatively, the same O_2^{q} molecule was optimized at coupled clusters with single and double excitation (CCSD) levels of theory (Scuseria & Schaefer, 1989) with the same basis sets. The stability of the optimized structures was tested by vibrational analysis (no imaginary vibrations). The Gaussian16 program suite (Frisch et al., 2016) was used for all quantum chemical calculations. The electronic structure analysis in terms of quantum theory of atoms in molecules (QTAIM) (Bader, 1994) was performed in the AIMAll package (Keith, 2016) using the wavefunctions from the Gaussian16 wfn and/ or fchk files. The *d*-electron populations at titanium were obtained using natural population analysis (Carpenter & Weinhold, 1988) as implemented in Gaussian16 (Frisch et al., 2016).

The domain averaged Fermi holes (DAFH) analysis (Ponec, 1998, 2010; Ponec & Cooper, 2007; Baranov *et al.*, 2012) was performed with the *D*-*Grid* package (version 5.1 Kohout, 2019) to analyze the bonding interactions between the central titanium atom and the O_2 moiety. DAFH eigenvectors have been visualized with the *IQmol* package (Gilbert, 2020).

2.4. (QT)AIM analysis

The total electron densities obtained from the multipole refinement and alternatively from theoretical calculations have been analyzed within the framework of the (QT)AIM (Bader, 1994). The results were evaluated in terms of atomic charges obtained using the electron density integrated over atomic basins and bond characteristics in terms of electron density ρ at bond critical points (BCPs) corresponding to saddle points at bond paths between individual atoms, its Laplacian $\nabla^2 \rho$ can be expressed by

$$\nabla^2 \rho = \lambda_1 + \lambda_2 + \lambda_3, \tag{2}$$

and bond ellipticity ε

$$\varepsilon = \lambda_1 / \lambda_2 - 1, \tag{3}$$

where $\lambda_1 < \lambda_2 < 0 < \lambda_3$ are the eigenvalues of the electron density Hessian at BCPs. Ring critical points are saddle points with $\lambda_1 < 0 < \lambda_2 < \lambda_3$ and cage critical points are local minima $(0 < \lambda_1 < \lambda_2 < \lambda_3)$ of electron density.

The BCP electron density (ρ_{BCP}) is proportional to the bond strength; the value and sign of its Laplacian ($\nabla^2 \rho_{BCP}$) describes the relative electron density contribution of the bonded atoms to the bond (covalent versus dative bonding); its bond ellipticity (ρ_{BCP}) describes its deviation from cylindrical symmetry (such as in ideal single or triple bonds) due to its double-bond character, mechanical strain and/or other perturbations.

2.5. DAFH analysis

In the case of the DAFH analysis, one can distinguish electron pairs (eigenvectors) which are broken or retained in a chosen part (domain) of a studied system (Ponec, 1998; Ponec & Cooper, 2007; Ponec et al., 2010; Baranov et al., 2012). In our case, we choose the O_2 moiety as the domain (using QTAIM atomic basins) to quantify the strength of dative interactions between Ti and O₂. Each DAFH eigenvector has an assigned eigenvalue (occupation) in the range 0-2, which represents the amount of electron density inside the chosen domain. Eigenvalues close to 2 correspond to a retained non-bonding electron pair within the domain, whereas eigenvalues below 2 represent that some part of the electron density is outside the domain due to bonding or dative interactions (and possibly also antibonding ones). For instance, an eigenvalue of 1.6 suggests a dative interaction, with 1.6 out of 2 electrons of a given electron pair (DAFH eigenvector, Fermi hole) being the part of the chosen domain. Eigenvectors with eigenvalues below 0.2 or 0.1 are mostly excluded from consideration, representing rather a numerical noise of the method. In addition, DAFH analysis uses an isopycnic localization (Cioslowski, 1990) to provide more useful information with respect to chemical intuition.

3. Results and discussion

3.1. Structure description

The coordination polyhedron of the central titanium atom, described by chromophore $[Ti(O_2)O_2N_2]$, is a deformed tetragonal pyramid (Fig. 1) with the peroxo anion in the apical position (*side-on* mode) and two oxygen and two nitrogen atoms in the basal plane. The titanium atom is shifted from the plane defined by basal plane atoms O(1), O(2), N(1) and N(2)

by 0.671 Å towards the peroxo anion. The angle between the basal plane and the plane defined by atoms O(1), O(2) and Ti is 86.45°. There are no similar crystal structures in the Cambridge Structural Database (Groom et al., 2016) except one (Guilard *et al.*, 1978) with the chromophore $[Ti(O_2)N_4]$ (side-on bonding mode). The titanium central atom is above the basal plane by 0.621 Å and the angle between the corresponding planes is 89.09°, defined above (Guilard et al., 1978). The interatomic distances in this compound are 1.822 (4), 1.827 (4) and 1.445 (5) Å, and in the title compound are 1.8699 (12), 1.8813 (11) and 1.5018 (16) Å for the Ti-O(1), Ti-O(2) and O(1)-O(2) bonds, respectively. It should be mentioned that the IAM model SHELXL refinement of the title compound gives the above values of 1.8522 (7), 1.8685 (6) and 1.4674 (9) Å, respectively. The O(1) - O(2) bond distances from the IAM model for both crystal structures points to an inadequacy of the model when comparing against the MM derived distance. On the other hand, the O(1) - O(2)bond distance from the MM refinement corresponds well with the value 1.499 (2) Å found in the solid sodium peroxide hydrate in which each oxygen atom is surrounded by four hydrogen bonds (Hill et al., 1997). The title crystal structure is stabilized by four intramolecular and two intermolecular hydrogen bonds (Table S1). It is important to state that interatomic distances from AIM refinement could suffer from systematic errors and the MM refinement obtains more accurate values. The interatomic distances and angles are shown in Table S2.

The *MM* refinement achieved a significant improvement of the agreement between the experimental and calculated structure factors when compared with ordinary IAM structure refinement. Furthermore, the accuracy in the interatomic distances is increased by an order of magnitude compared with a routine IAM *SHELXL* refinement.

3.2. Topology of MM and DFT charge densities

The aim here was to characterize the studied crystal structure and topological properties of the *MM*-refined experimental charge density and to make a comparison with density functional theory (DFT). A further task was to detect the amount of electron density transfer from the peroxide anion to the rest of the molecule and thus characterize the changes of the peroxo O-O bond due to the coordination to Ti(IV).

According to the QTAIM BCP descriptors obtained for the *MM*-refined charge density, the strongest coordination bonds are Ti-O(1) and Ti-O(2) where ρ_{BCP} and $\nabla^2 \rho_{BCP}$ have the highest values (Table 2). Ti-O(3) and Ti-O(4) bonds from phenylenediamine are weaker and the coordination bonds Ti-N(1) and Ti-N(2) from vanillin are the weakest. This is also confirmed in the theoretical results (see Tables 2, S2 and S3). All experimental and theoretical values of Ti-O and Ti-N coordination bonds are in good agreement with each other except the ellipticity of the Ti-O(1) and Ti-O(2) bonds in the peroxo anion. The experimental ellipticity of Ti-O(1) and Ti-O(2) bonds are about ten times higher than the theoretical ones. Such high values of ellipticity are even larger than



Figure 1

ORTEP plot of the title compound. Thermal ellipsoids are drawn at 30% probability. Symmetry codes used: (i) x, -1 + y, z; (ii) 1 - x, 1 - y, 1 - z.

the experimental values for cyclopropane which are in the interval 0.61-0.67 (Bacsa & Briones, 2013). Ellipticity expresses the amount of π bonding contributions as well as the mechanical strain at a particular bond in cyclic structures. The MM BCP Laplacian values for the coordination bonds are found to be larger than the theoretical values, especially for the oxygen atoms, in particular O(1) and O(2), see Table 2. A similar statement is true for the O(1) - O(2) BCP Laplacian as well as the BCP electron density, see Table 2. Interestingly, the O(1) - O(2) BCP Laplacian is found to be positive, which is not consistent with a typical covalent bond. The explanation could be based on the shift of the electron density in the region between the two oxygen atoms to the titanium atom, and thus the less negative region on the oxygen nucleus interaction with the negative region which resembles the closed-shell interaction. The difference between the theoretical Laplacian of the free O_2^{2-} anion [Fig. 2(c)] and the MM Laplacian of the coordinated O_2^{2-} anion is that the first is symmetrical according to the center of the O-O bond, and in the case of the coordinated one, the valence shell charge concentration (VSCC) is asymmetric and shifted to the central titanium atom [Figs. 2(a) and 2(b)].

In our last paper (Vénosová *et al.*, 2020) we tested the improvement of *MM* flexibility by improving the radial functions of sulfur, oxygen and nitrogen atoms according to the work by Dominiak & Coppens (2006). Zeta values were taken from the *JANA2006* database (Petříček *et al.*, 2014). For the $C_{22}H_{18}N_2O_6Ti \cdot H_2O$ complex under study, we considered the radial function flexibility of the oxygens, but no changes were observed.

In the case of QTAIM charges, the transfer of charge density from the O(1)-O(2) moiety is found to amount to almost two electrons in the *MM*-refined results (1.61 e), whereas in the theoretical results we found a charge transfer of only one electron (0.98 e) from the O(1)-O(2) moiety, see Table S4. Still the *MM* and DFT QTAIM charges of Ti are both close to two. Further differences in the *MM* and DFT charges are found also for the remaining oxygen atoms, with

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Table 2 Selected AIM electron density properties at bond critical points (bond path length $d_{12} = d_1 + d_2$).

Geometry	Method	Bond		BCP characteristics					
		Atom 1	Atom 2	$d_{12}({ m \AA})$	$\rho_{\rm BCP} ({ m e}{ m \AA}^3)$	$ abla^2 ho_{ m BCP}$ (e Å ⁵)	ε	d_1 (Å)	d_2 (Å)
Exp.	Exp.	Ti	O(1)	1.8777	0.88 (2)	18.73 (4)	0.81	0.9553	0.9224
Exp.	Exp.†	Ti	O(1)	1.8795	0.89 (2)	19.04 (4)	1.03	0.9474	0.9320
Exp.	DFT	Ti	O(1)	1.8764	0.9161	11.6452	0.073	0.9519	0.9245
DFT	DFT	Ti	O(1)	1.8272	1.0378	12.8197	0.086	0.9296	0.8976
Exp.	Exp.	Ti	$O(2)^{*}$	1.8885	0.87 (2)	17.64 (4)	0.82	0.9576	0.9308
Exp.	Exp.†	Ti	$O(2)^{*}$	1.8922	0.82 (2)	17.00 (4)	1.05	0.9489	0.9434
Exp.	DFT	Ti	$O(2)^{*}$	1.8884	0.8823	11.4024	0.052	0.9566	0.9318
DFT	DFT	Ti	$O(2)^{*}$	1.8451	0.9835	12.8530	0.059	0.9372	0.9079
Exp.	Exp.	Ti	O(3)	1.8966	0.79(1)	15.95 (4)	0.10	0.9545	0.9421
Exp.	Exp.†	Ti	O(3)	1.8989	0.82(1)	13.92 (3)	0.13	0.9556	0.9433
Exp.	DFT	Ti	O(3)	1.9009	0.8122	13.6067	0.057	0.9565	0.9444
DFT	DFT	Ti	O(3)	1.9101	0.7439	12.3081	0.061	0.9599	0.9502
Exp.	Exp.	Ti	O(4)	1.9235	0.69(1)	14.59 (3)	0.19	0.9739	0.9496
Exp.	Exp.†	Ti	O(4)	1.9262	0.71 (1)	13.63 (3)	0.20	0.9745	0.9517
Exp.	DFT	Ti	O(4)	1.9275	0.7689	12.9308	0.032	0.9737	0.9538
DFT	DFT	Ti	O(4)	1.9560	0.6732	11.1183	0.026	0.9871	0.9689
Exp.	Exp.	O(1)	O(2)	1.5091	2.45 (3)	16.16 (4)	0.06	0.7701	0.7390
Exp.	Exp.†	O(1)	O(2)	1.5118	2.32 (2)	11.93 (6)	0.05	0.7638	0.7479
Exp.	DFT	O(1)	O(2)	1.5052	1.5914	5.2606	0.070	0.7526	0.7523
DFT	DFT	O(1)	O(2)	1.4595	1.8050	3.9597	0.071	0.7303	0.7292

† Two different scattering factors for oxygen atoms.

the *MM* charges being more negative. The more negative the *MM* charges of O(5) and O(7) atoms counterbalance the larger charge transfer from the O(1)–O(2) moiety, when comparing to DFT QTAIM charges. Experimental results take into account intermolecular hydrogen bonds and non-covalent interactions (Table S5) from adjacent molecules which could stabilize a larger charge transfer from the O(1)–O(2) moiety, whereas DFT results (see below) reflect the electronic structure of an isolated molecule.

Furthermore, we will consider the MM deformation density [Fig. 3(a)] and Laplacian for the Ti-O(1)-O(2) plane for comparison with the QTAIM results [Fig. 2(a) and 2(b)]. A classical coordination bond is understood as a bond where the lone electron pair of a single donor atom and the depopulated d-orbital of the central atom are involved. In the studied complex, one finds a perpendicular coordination of the O₂ moiety with respect to Ti. Hence, MM deformation densities in the Ti-O(1)-O(2) plane uncover a superposition of a σ -like interaction in the O₂ moiety, with the coordination bond contributions between the depopulated d(Ti) orbitals and the fully occupied $\pi(O_2)$ and $\pi^*(O_2)$ orbitals, see Fig. 3(*a*). Thus, the deformation O(1)-O(2) bonding density appears to be shifted towards Ti. The Laplacian for the Ti-O(1)-O(2)plane shows also that the O(1) and O(2) maxima are tilted off the actual O(1) - O(2) axis and build a VSCC towards titanium on each oxygen atom. This Laplacian map also confirms the positive Laplacian region between the O(1) - O(2) atoms (Table 2).

Experimentally found VSCCs consistently found in the figures of electrostatic potential (BCP) are placed outside the triangle Ti(1)-O(1)-O(2), the gradient field trajectory plot of electrostatic potential, the static deformation map and the map of Laplacian [Fig. 3(*b*)].

Last but not least, Table S6 presents the MM and DFT dorbital populations. Generally, the individual MM d-orbital populations are higher by approximately 0.3 e⁻ than those of DFT (except $d_{x^2-y^2}$ and d_{xz}). Hence the total *MM d*-population is higher by one electron than in the case of DFT. Still, the experimental MM population is relevant to multipole moment decomposition and should lead to a Ti charge of one, while the MM OTAIM charge of Ti is two. In the case of DFT d-orbital natural populations, the sum of these *d*-populations is close to 2.0, although the Ti charge from DFT based natural population analysis is 1.24 e⁻. We have also optimized the local coordinate system for the titanium atom by minimizing the MM populations of d_{z^2} and $d_{x^2-y^2}$ orbitals $(d_{z^2} + d_{x^2-y^2})$ using the program ERD (Sabino & Coppens, 2002). The obtained $d_{x^2-v^2}$ orbital population was 0.5113 e⁻ [the Ti-O(3), Ti-O(4), Ti-N(1) and Ti-N(2) coordination bonds], the d_{xy} orbital population was 0.7351 e⁻ (non-bonding orbital) and d_{z^2} orbital population was 0.7086 e⁻ (axial interaction with the peroxide anion).

3.3. Theoretical assessment of Ti-peroxo interactions

First of all, we focused on the O₂ moiety itself. One can compare the O–O QTAIM bond descriptors in the title compound with references such as O₂, O₂⁻ and O₂²⁻, see Table S7 (O₂²⁻ in the triplet spin state is unstable). We may conclude that the B3LYP and CCSD data are in a reasonable agreement. The bond length increases with the negative charge q of the O₂^q system, and hence the BCP electron density (*i.e.* bond strength) decreases. This is in line with occupying the $\pi^*(O_2)$ orbitals, with O₂²⁻ having fully populated π and π^* MOs. Actually, the O(1)–O(2) bond length in the complex studied is in between the O–O bond length of the



(a) Experimental Laplacian distribution $L(r) \simeq \nabla^2 \rho(r)$ in the Ti(1)– O(1)–O(2) plane; (b) theoretical Laplacian distribution $L(r) \simeq \nabla^2 \rho(r)$ in the Ti(1)–O(1)–O(2) plane; (c) theoretical Laplacian of the free O₂^{2–} anion. Contours are drawn at -1.0×10^{-3} , $\pm 2.0 \times 10^{n}$, $\pm 4.0 \times 10^{n}$, $\pm 8.0 \times 10^{n}$ (n = -3, -2 - 1, 0, +1, +2 + 3) e Å⁻⁵, with positive contours drawn with a solid blue line and negative contours with a dashed red line.

free O_2^- and O_2^{2-} systems, compare Tables 2 and S7. Similar relations hold for BCP electron density and the Laplacian, although the O(1)-O(2) BCP Laplacian in the complex under study is closer to O_2^{2-} . Hence, upon Ti^{IV} $\leftarrow O_2^-$ coordination, the charge transfer from the $\pi(O_2)$ and $\pi^*(O_2)$ orbitals leads to shortening of the O-O distance because of a lower repulsion between lower charge densities at particular oxygen atoms. Note also that the BCP Laplacian in ${}^{1}O_2{}^{2-}$ is positive, at both CCSD and B3LYP levels of theory.

To obtain further insight into the $Ti^{IV} \leftarrow O_2^{2-}$ coordination in the complex under study, DAFH analysis was performed, defining O_2 as the domain to inspect the bonding interactions (DAFH eigenvectors) which are retained (eigenvalues close to 2) or split (eigenvalues <2, but >0.05) because of the domain choice itself. In the case of the O₂ domain, one finds nine such DAFH eigenvectors. Four of the DAFH eigenvectors can be assigned to the 1s- and 2s-like densities on the oxygens (eigenvalues > 1.98), we will exclude these from consideration. Instead, we show the three distinguished contributions, *i.e.* the $\sigma_{O(1)-O(2)}$ and the p_z O(1) and the p_y O(1)-like DAFH eigenvectors, see Fig. 4. DAFH p_z - and p_y -like eigenvectors of O(2) and O(1) are similar, hence those of O(2) are not shown in any detail. From the eigenvalues and pictorial representation of DAFH eigenvectors, we may conclude that the $\sigma_{O(1)}$ O(2) bond does not really contribute outside the peroxo-like moiety. The double-occupied $p_{y}(O)$ orbitals [perpendicular to the Ti - O(1) - O(2) plane] contribute less to the coordination bond than the $p_z(O)$ orbitals [in the Ti-O(1)-O(2) plane]. Apparently, DAFH analysis breaks the $Ti^{IV} \leftarrow O_2^{2-}$ interactions into individual $Ti^{IV} \leftarrow O(1)$ and $Ti^{IV} \leftarrow O(2)$ eigenvectors, which have a small π -like O(1)–O(2) contribution, but which are in different phases for the O(1) and O(2) atoms. Hence instead of the formal $\pi_{y,z}$ and $\pi^*_{y,z}$ orbitals we obtained nonbonding p_v and p_z orbitals in the DAFH representation which mediate the dative interactions (coordination bonds) between Ti and O(1)–O(2). The polarization of the $\sigma_{O(1)-}$ O(2) DAFH eigenvector and the dative character of $p_z(O)$ DAFH eigenvectors [Figs. 4(a) and 4(c)] are in line with the shape of the MM deformation density and Laplacian maps being shifted or pointed from O(1) - O(2) towards Ti (Fig. 2).

4. Conclusions

By means of the charge density study presented here we proved that, in the title compounds, the O–O bonding electron density is significantly shifted towards the central titanium atom. The BCP Laplacians of the O–O bonds in the O₂ molecule, as well as in the title molecule are positive. This indicates that VSCC belongs to the area which is outside the O–O bond. The difference is that the VSCC in peroxocomplexes is asymmetric with respect to the O–O bond and symmetric for the O₂ molecule. The O–O bond in the peroxide complex is weakened and therefore could be susceptible to a nucleophilic addition reaction. The differences between the experimental and theoretical electronic structure clearly show that, in the case of the stabilizing effect of molecules in the surroundings, this attenuating effect of O–O binding could be increased. Properties of the O–O moiety

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Figure 3

(a) Static electron deformation density in the plane defined by the atoms Ti(1) - O(1) - O(2). Contour spacing 0.1 e Å⁻³, with positive contours drawn with a solid blue line and negative contours with a dashed red line; (b) gradient field trajectory plot of electrostatic potential in the plane Ti - O(1) - O(2); (c) experimental electrostatic potential on the 3D-isosurface of the experimental electron density (0.3 e Å⁻³); (d) three-dimensional plot (Hübschle & Dittrich, 2011) of the Laplacian of electron density around Ti at an isosurface value of 80 e Å⁻⁵.

could be modified by suitable surrounding of the central titanium atom. In the case of different central atoms, the behavior of similar complexes could be comparable. Shifting the electron density is just a first step in the chain of subsequent reaction mechanisms. By modifying the supporting ligand, both electrophilic and nucleophilic reactions can take place. To the best of our knowledge, this is the first example of a charge density study of a coordination compound in which a peroxo anion is bonded to a 3*d* central atom. Interestingly, titanium has been found in a number of marine organisms which makes this metal important from a biological point of view. In fact, several titanium compounds have been shown to possess anticancer properties, enzyme inhibiting and antibacterial activities. Budo-titane and titanocene dichloride have been used in human anticancer clinical trials, however, to

date have not reached clinical use. The main problem with the use of these compounds is the dose limit toxicity and solubility. Recently Obeid *et al.* (2012) reported Schiff-base titanium(IV) complexes with promising anticancer and antibacterial properties. In this work, authors proposed that DNA cleavage activity of Ti(IV) complexes was achieved *via* ROS-induced oxidative damage, predominantly by the DNA damaging activity of the hydroxyl radical. The Schiff base Ti(IV) peroxo complex studied in this work may exhibit similar biological (anticancer) properties, which may be enhanced by the presence of a peroxo group capable of participating in freeradical DNA damaging cascades. Related studies are underway. Differences between experimental and theoretical results, in which the properties of the isolated molecule and the molecule in the crystal are similar, are a good inspiration



Figure 4

(a) σ O(1)–O(2)-like DAFH eigenvector (1.95); (b) p_y O(1)-like DAFH eigenvector (1.82); (c) p_z O(1)-like DAFH eigenvector (1.64) (DAFH eigenvalues are shown in parentheses).

to improve the model of the molecular system. A theoretical study of the modified environment of the donor atoms can be used to tune the nature of the O-O bond. This electronic structure could be used as a possible model structure of the reaction center for hemoglobin or other metalloproteins.

5. Related literature

The following reference is cited in the supporting information: Allen & Bruno (2010).

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References

- Abrahams, S. C. & Keve, E. T. (1971). Acta Cryst. A27, 157-165.
- Allen, F. H. & Bruno, I. J. (2010). Acta Cryst. B66, 380-386.
- Bacsa, J. & Briones, J. (2013). Acta Cryst. C69, 910-914.
- Bader, R. F. W. (1994). Atoms in Molecules: a Quantum Theory. Oxford University Press.
- Baranov, A. I., Ponec, R. & Kohout, M. (2012). J. Chem. Phys. 137, 214109.
- Becke, A. D. (1988). Phys. Rev. A38, 3098-3100.
- Carpenter, J. E. & Weinhold, F. (1988). J. Mol. Struct. Theochem, 169, 41-62.
- Cioslowski, J. (1990). Int. J. Quantum Chem. Symp. 38, 15-28.
- Coppens, P. (1997). X-ray Charge Densities and Chemical Bonding. Oxford University Press.
- Dakanali, M., Kefalas, E. T., Raptopoulou, C. P., Terzis, A., Voyiatzis, G., Kyrikou, I., Mavromoustakos, T. & Salifoglou, A. (2003). *Inorg. Chem.* 42, 4632–4639.
- De Lile, J. R., Heine, T. & Zhou, S. (2017). Comput. Mat. Sci. 129, 24–36.
- Deng, Y. F. & Zhou, Z. H. (2008). Inorg. Chem. Commun. 11, 1064– 1066.
- DiPasquale, A. G., Kaminsky, W. & Mayer, J. M. (2002). J. Am. Chem. Soc. **124**, 14534–14535.
- Dolomanov, O. V., Bourhis, L. J., Gildea, R. J., Howard, J. A. K. & Puschmann, H. (2009). J. Appl. Cryst. 42, 339–341.
- Dominiak, P. M. & Coppens, P. (2006). Acta Cryst. A62, 224-227.
- Farrugia, L. J. (2012). J. Appl. Cryst. 45, 849-854.
- Frisch, M. J., Trucks, G. W., Schlegel, H. B., Scuseria, G. E., Robb, M. A., Cheeseman, J. R., Scalmani, G., Barone, V., Petersson, G. A., Nakatsuji, H., Li, X., Caricato, M., Marenich, A. V., Bloino, J., Janesko, B. G., Gomperts, R., Mennucci, B., Hratchian, H. P., Ortiz, J. V., Izmaylov, A. F., Sonnenberg, J. L., Williams-Young, D., Ding, F., Lipparini, F., Egidi, F., Goings, J., Peng, B., Petrone, A., Henderson, T., Ranasinghe, D., Zakrzewski, V. G., Gao, J., Rega, N., Zheng, G., Liang, W., Hada, M., Ehara, M., Toyota, K., Fukuda, R., Hasegawa, J., Ishida, M., Nakajima, T., Honda, Y., Kitao, O., Nakai, H., Vreven, T., Throssell, K., Montgomery, J. A. Jr, Peralta, J. E., Ogliaro, F., Bearpark, M. J., Heyd, J. J., Brothers, E. N., Kudin, K. N., Staroverov, V. N., Keith, T. A., Kobayashi, R., Normand, J., Raghavachari, K., Rendell, A. P., Burant, J. C., Iyengar, S. S., Tomasi, J., Cossi, M., Millam, J. M., Klene, M., Adamo, C., Cammi, R., Ochterski, J. W., Martin, R. L., Morokuma, K., Farkas, O., Foresman, J. B. & Fox, D. J. (2016). Gaussian16, Revision B. 01, Gaussian, Inc., Wallingford CT.
- Fukin, G. K., Baranov, E. V., Cherkasov, A. V., Rumyantcev, R. V., Artemov, A. N. & Sazonova, E. V. (2019). *Russ. J. Coord. Chem.* 45, 680–688.
- Gao, Ch., Macetti, G. & Overgaard, J. (2019). *Inorg. Chem.* 58, 2133–2139.
- Gilbert, A. (2020). *IQmol* version 2.14. Australian National University ACT. http://www.iqmol.org.

research papers

- Grimme, S., Antony, J., Ehrlich, S. & Krieg, H. (2010). J. Chem. Phys. 132, 154104.
- Groom, C. R., Bruno, I. J., Lightfoot, M. P. & Ward, S. C. (2016). Acta Cryst. B72, 171–179.
- Guilard, R., Latour, L. M., Lecomte, C., Marchon, J. C., Protas, J. & Ripoll, D. (1978). *Inorg. Chem.* **17**, 1228–1237.
- Hansen, N. K. & Coppens, P. (1978). Acta Cryst. A34, 909-921.
- Herich, P., Bučinský, L., Breza, M., Gall, M., Fronc, M., Petřiček, V. & Kožíšek, J. (2018a). Acta Cryst. B74, 681–692.
- Herich, P., Kucková, L., Moncol, J. & Kožíšek, J. (2018b). Z. Kristallogr. 233, 745–752.
- Hill, G. S., Holah, D. G., Kinrade, S. D., Sloan, T. A., Magnuson, V. R. & Polyakov, V. (1997). *Can. J. Chem.* **75**, 46–51.
- Hübschle, C. B. & Dittrich, B. (2011). J. Appl. Cryst. 44, 238-240.
- Keith, T. A. (2016). *AIMALL*, version 16.05.18. TK Gristmill Software, Overland Park KS, USA. http://aim.tkgristmill.com.
- Kohout, M. (2019). *DGrid*, version 5.1. Dresden. http:// www.cpfs.mpg.de/~kohout/dgrid.html.
- Kožíšek, J., Hansen, N. K. & Fuess, H. (2002). Acta Cryst. B58, 463– 470.
- Lee, C., Yang, W. & Parr, R. (1988). Phys. Rev. B, 37, 785-789.
- Meindl, K. & Henn, J. (2008). Acta Cryst. A64, 404-418.
- Niepötter, B., Herbst-Irmer, R. & Stalke, D. (2015). J. Appl. Cryst. 48, 1485–1497.
- Obeid, A., El-Shekeil, A., Al-Aghbari, A. & Al-Shabi, J. (2012). J. Coord. Chem. 65, 2762–2770.
- Petříček, V., Dusek, M. & Palatinus, L. (2014). Z. Kristallogr. 229, 345–352.
- Ponec, R. (1998). J. Math. Chem. 23, 85-103.
- Ponec, R., Bučinský, L. & Gatti, C. (2010). J. Chem. Theory Comput. 6, 3113–3121.
- Ponec, R. & Cooper, D. L. (2007). J. Phys. Chem. A, 111, 11294– 11301.
- Rebilly, J.-N., Colasson, B., Bistri, O., Overa, D. & Reinaud, O. (2015). *Chem. Soc. Rev.* **44**, 467–489.

- Sabino, J. R. & Coppens, P. (2002). Acta Cryst. A59, 127-131.
- Scatena, R., Guntern, Y. T. & Macchi, P. (2019). J. Am. Chem. Soc. 141, 9382–9390.
- Schmøkel, M. S., Bjerg, L., Larsen, F. K., Overgaard, J., Cenedese, S., Christensen, M., Madsen, G. K. H., Gatti, C., Nishibori, E., Sugimoto, K., Takata, M. & Iversen, B. B. (2013). Acta Cryst. A69, 570–582.
- Scuseria, G. E. & Schaefer, H. F. III (1989). J. Chem. Phys. 90, 3700-3703.
- Sheldrick, G. M. (2015a). Acta Cryst. A71, 3-8.
- Sheldrick, G. M. (2015b). Acta Cryst. C71, 3-8.
- Stephens, P. J., Devlin, F. J., Chabalowski, C. F. & Frisch, M. J. (1994). J. Phys. Chem. 98, 11623–11627.
- Stoe & Cie (2018). X-AREA Integrate. Stoe & Cie GmbH, Darmstadt, Germany.
- Su, Z. & Coppens, P. (1998). Acta Cryst. A54, 646-652.
- Tengvall, P., Wälivaara, B., Westerling, J. & Lundström, I. J. (1991). J. Colloid Interface Sci. 143, 589–592.
- Valko, M., Izakovic, M., Mazur, M., Rhodes, C. J. & Telser, J. (2004). Mol. Cell. Biochem. 266, 37–56.
- Valko, M., Klement, R., Pelikan, P., Boca, R., Dlhan, L., Boettcher, A., Elias, H. & Mueller, L. (1995). J. Phys. Chem. 99, 137– 143.
- Valko, M., Morris, H. & Cronin, M. T. D. (2005). Curr. Med. Chem. 12, 1161–1208.
- Vénosová, B., Koziskova, J., Kožíšek, J., Herich, P., Lušpai, K., Petricek, V., Hartung, J., Müller, M., Hübschle, C. B., van Smaalen, S. & Bucinsky, L. (2020). Acta Cryst. B76, 450–468.
- Volkov, A., Macchi, P., Farrugia, L. J., Gatti, C., Mallinson, P., Richter, T. & Koritsanszky, T. (2016). XD2016. http://www.chem.gla.ac.uk/ ~louis/xd-home/.
- Vosko, S. H., Wilk, L. & Nusair, M. (1980). Can. J. Phys. 58, 1200–1211.
- Zhou, Z. H., Deng, Y. F., Liu, Q. X., Zhang, H. L., Mak, T. C. W. & Chow, Y. L. (2007). *Inorg. Chem.* **46**, 6846–6848.